### Houston Community College and the West Houston Center for Science & Engineering

### Presents

The 2017 Research Experiences and Exploration in Materials Science (REEMS) Research Experience for Undergraduates<sup>1</sup> Poster Presentation & Recognition Ceremony<sup>2</sup>



August 10, 2017 Held at the MD Anderson Cancer Center UT Health Graduate School of Biomedical Sciences

<sup>1</sup> Funding for this event was provided by the National Science Foundation, Division of Materials Research, DMR 1460564 and the HCC West Houston Center for Science and Engineering. Special thanks to the administrators and staff at the Graduate School of Biomedical Sciences for providing the venue for this event.

<sup>2</sup> Second Revision, September 14, 2017

### The 2017 Research Experiences and Exploration in Materials Science (REEMS): Research Experience for Undergraduates

### **Poster Presentation & Recognition Ceremony**

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### The 2017 Research Experiences and Exploration in Materials Science (REEMS): Research Experience for Undergraduates

### **Poster Presentations & Recognition Ceremony**

### August 10, 2017

### Schedule of Events

8:00 – 8:45 am	Student arrival and poster set-up
9:00 am	Judges Arrive
9:45 – 11:30 am	Poster presenters stand at posters for judging
Noon – 12:45 pm	Lunch
1:00 – 2:30 pm	Poster Awards and Recognition Ceremony



### Houston Community College 2017 Summer Undergraduate Research Program in Materials Science

NSF REU: Research Experiences and Exploration in Materials Science (REEMS) for Houston Community College Science and Engineering Students (NSF DMR 1460564)

Bartlett (Bart) M. Sheinberg serves as the Principal Investigator for this award.

In 2015, The West Houston Center for Science and Engineering (WHC) was awarded funding from the National Science Foundation to develop an innovative materials science educational program and summer research program (REU) specifically for Houston Community College students.

The REEMS program, inclusive of the REU, provides a series of multidisciplinary learning experiences in chemistry, physics, engineering, the biological sciences, medicine, computational science, economics, and public policy. The Program experience includes individualized transfer guidance and academic advising to insure that each of the REEMS students transfer in a timely manner to complete their undergraduate education, an overview of graduate school as the next step in their academic careers, and student networking opportunities with materials professionals.

REEMS REU students represent diverse cultural, economic, educational backgrounds, and ages. While their backgrounds, and academic-career goals may vary, each student has an intense motivation to learn and a willingness to explore new challenges. Program evaluation data presents a strong correlation between student participation in the REEMS program with an enhanced appreciation of mathematical, physical & biological sciences, computer programming and engineering concepts discussed in their coursework. Students quickly grasp the importance and relevance of their coursework as they apply these concepts to solve research problems and begin to think about how their education and experiences in materials science can be applied to solving societal technology and material challenges.

The 2017 REEMS REU poster session provided the opportunity for students to share their summer research experiences with other students, university faculty, and the public. This book contains an overview of 2017 REEMS REU student profiles, research abstracts and posters; information on each of the REEMS Research faculty; profiles of the REEMS REU poster session judges; and recognition of the West Houston staff and Advisory Council members.

### 2017 REEMS REU Students

#### **Yasaman Adel**



#### yasamanadel@gmail.com

REEMS REU Faculty Supervisor: Dr. Laura Smith Callahan, McGovern Medical School, UT Health Science Center – Houston

REEMS REU Mentor: Dr. Xi Lu

Transferring in fall 2017 to the University of Houston –Biological Sciences

#### Brenda Arceneaux



#### ba.arcen@yahoo.com

REEMS REU Faculty Supervisor: Dr. James K. Meen, University of Houston

REEMS REU Mentor: Dr. Karoline Müller

Returning to HCC in the Fall of 2017 to continue coursework in engineering and serving as an adjunct HCC instructor.

### **Andrew Michael Catalanotto**



#### am.catala67@gmail.com

REEMS Faculty Supervisor Advisor: Dr. Zachary Cordero, Rice University

**REEMS REU Mentor: Logan Ware** 

Transferring to the University of Texas – Tyler (Houston) in the fall of 2017 to study Electrical Engineering

#### Joshua Jackson



#### joshnm28@gmail.com

REEMS REU Faculty Supervisor: Dr. Rafael Verduzco, Rice University

REEMS REU Mentor: Jorge Wu Mok

Transferring to Prairie View A&M University in fall of 2017 to study electrical engineering

Page **O** 

#### **Frank Kornet**



Frank Kornet (center), Margaret Cheung (right) & Millad Ghane (left)

#### frkornet@gmail.com

REEMS REU Faculty Supervisor: Dr. Margaret Cheung, University of Houston & Rice University

**REEMS REU Mentor: Millad Ghane** 

Returning to HCC in fall of 2017 to prepare for graduate school.

#### Monica R. Martinez



martinezhenriquez@hotmail.com

REEMS REU Faculty Supervisor – Dr. James K. Meen, University of Houston

REEMS REU Mentor: Dr. Karoline Müller

Transferring fall 2017 to University of Houston, Honors College, to study Mechanical Engineering

### Rodrigo Munoz-Zarruk



rodrigo.munozzarruk@mavs.uta.edu

REEMS REU Faculty Supervisor: Dr. Rafael Verduzco, Rice University

**REEMS REU Mentor: Jorge Wu Mok** 

Transferring to the University of Texas - Arlington in the fall of 2017 to study mechanical engineering

### Tasmia Nadeem



#### tasmia.nadeem27@gmail.com

REEMS REU Faculty Supervisor: Dr. Laura Smith -Callahan-McGovern Medical School, University of Texas Health Science Center –Houston

REEMS REU Mentor- Dr. Xi Lu.

Transferring fall of 2017 to attend the University of Texas MD Anderson School of Public Health, Bachelors Program in Molecular Genetics.

### Gelareh Nobakht



#### gelareh.no@gmail.com

REEMS REU Faculty Supervisor: Dr. Jakoah Brgoch, University of Houston

REEMS REU Mentor: Aria Mansouri Tehrani

Returning to HCC in the fall of 2017 and transferring to the University of Texas-Dallas, spring 2018 to study computer science

### **Nathaniel Ocanas**



#### nathan\_ocanas@yahoo.com

REEMS REU Faculty Supervisor – Dr. Zachary Cordero, Rice University

REEMS REU Mentor: - Mr. Austin A. Ward

Transferring fall 2017 to study electrical engineering at the University of Texas – Tyler (Houston)



#### Zeshan Rizvi



#### shanoo824@gmail.com

REEMS REU Faculty Supervisor: Dr. Jakoah Brgoch, University of Houston

REEMS REU Mentor: A. Oliynyk

Transferring fall 2017 to Prairie View A&M University to study chemical engineering

### **Kevin Scholtes**



#### ktscholtes@gmail.com

REEMS REU Faculty Supervisor: Dr. Megan Robertson, University of Houston

**REEMS REU Mentor: - Dr. Wenyue Ding** 

Fall Semester: Recipient of an internship at the Department of Energy, Oak Ridge National Lab, Center for Nanophase Materials Sciences

Transferring spring 2018 to study electrical engineering at the University of Texas – Tyler (Houston)

### Brandon Vu



#### brandonvu00@gmail.com

REEMS REU Faculty Supervisor: Dr. James K. Meen, University of Houston

REU Mentor: - Dr. Karoline Müller

Returning to Houston Community College in the 2017/2018 academic year and transferring to Texas A&M University in the fall of 2018 to study electrical engineering. Brandon is a member of the Texas A&M Engineering Academy at Houston Community College.

### 2017 REEMS REU STUDENT ABSTRACTS

### 2017 REEMS REU POSTER PRESENTERS

STUDENT NAME	REEMS FACULTY SUPERVISOR	Poster #
Gelareh Nobakht	Jakoah Brgoch	1
Zeshan Rizvi	Jakoah Brgoch	2
Yasaman Adel	Laura Smith-Callahan	3a
Tasmia Nadeem	Laura Smith-Callahan	3b
Andrew Catalanotto	Zachary Cordero	4
Nathaniel Ocanas	Zachary Cordero	5
Frank Kornet	Margaret Cheung	6
Brenda Arceneaux	James Meen	7
Monica Martinez	James Meen	8
Brandon Vu	James Meen	9
Kevin Scholtes	Megan Robertson	10
Joshua Jackson	Rafael Verduzco	11
Rodrigo Munoz Zarruk	Rafael Verduzco	12

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### Temperature-induced Phase Transformation in NiTi – A Molecular Dynamics Study

<u>G.Nobakht</u><sup>2</sup>, A. Mansouri Tehrani<sup>1</sup>, J. Brgoch<sup>1</sup>

<sup>1</sup>Department of Chemistry, University of Houston, Houston, Texas <sup>2</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College, Houston, Texas

Nickel-titanium (NiTi) shape memory alloys are widely used in medical and industrial fields. The shape memory behavior is due to a phase transformation between high temperature B2 and low temperature B19' phase. Here, the temperature-induced phase transformation in NiTi shape memory alloys is investigated by Molecular Dynamics (MD) simulation. The quality of MD simulations is substantially dependent on the used interatomic potential. Therefore, we compare the implementation of two most common interatomic potentials for NiTi MD simulations, EAM (Embedded Atom Method) and MEAM (Modified Embedded Atom Method). A series of simulations were performed using LAMMPS (Large-scale Atomic/Molecular Massively Parallel Simulator) software. The temperature was controlled using the Nose-Hoover thermostat while Parrinello-Rahman method was used for pressure. After the initial relaxation, the system was cooled to 200 K. During cooling, the phase transformation between cubic B2 (austenite) and B19' (martensite) phase occurs at 302.18 K using EAM and 261.8 K using MEAM potential. At high temperature (~400K), NiTi has austenite structure, by decreasing the temperature to ~200K, the system transforms to martensite phase with  $\beta$  of 95.30° and 95.47° for EAM and MEAM, respectively. The subsequent heating to  $\sim$ 400K, results in the recovery of the austenite phase ( $\beta$ =90°). This study shows that both EAM and MEAM potentials are capable of reproducing temperature-induced phase transformation in NiTi and they are in good agreement with experimental data. However, development of more accurate, versatile interatomic potentials for this intriguing alloy is further encouraged.

This work is supported by the American Chemical Society Petroleum Research Fund (55625- DNI10), the National Science Foundation Division of Civil, Mechanical & Manufacturing Innovation (NSF CMMI 15-62142), and National Science Foundation Division of Materials Research (NSF DMR 14-60564)



# Z

# 77204

# atom method **Modified embedded** VS (EAM)

Phase

# $\phi_{lphaeta}(r_{ij}),$

The embedded-atom method is	describing the energy between atoms	(interatomic potential). The energy is a	function of sum of functions between	the atom and its neighbors. The EAM is	related to Finnis-Sinclair model, which is	formulated for FCC metallic alloys.	<ul> <li>spherical symmetry</li> </ul>	
	·- 7		_					

The MEAM potential behaves like the EAM potential but MEAM potential

- employs cubic splines instead of analytic functions.
- s, p and f symmetry. Ideal for modeling fcc,hcp and
- diamond cubic.
- angular forces + EAM MEAM

# B19'(martensite) with EAM and 2NN MEAM potential and NiTi phase transition between B2(austenite)

<i>γ[°]</i>	06	95.49	06	95.30	06	97.9
c [Å]	4.81	4.5	4.2	4.224	4.265	4.657
b [Å]	4.81	4.9	4.2	4.616	4.265	4.114
a [Å]	4.82	4.9	30.01	3.032	3.016	2.909

# boundary <u>conditions</u> <u>Periodic</u>

During the process the system was isothermal and isobaric(NPT) Simulation step: 0.0005 Ps

#### temperature (400K), in 309K with Shape N Phase tra cooling, the phas Molecular EA at 250 occurs Conclusion 200 K. 200 phase Temperature 400 06 93 92 91 96 95 94

MEAM, and

Tehrani, A. M.; Sha

Ko, W.-S.; Grabows 014901

cknowledgn

Dynamics simulations res memory mechani ~200K, the syster References

# transformation phase nsitic The marte

X Houston, of Houston, Brgoch University **—** Tehrani, Chemistry, Mansouri of Ŕ Department G.Nobakht,



![](_page_13_Picture_26.jpeg)

# Introduction

![](_page_13_Picture_29.jpeg)

MD

and molecules

#### Zeshan Rizvi

#### Machine-Learning-Driven Superhard Material Search: ReWC<sub>0.8</sub> and Mo<sub>2-x</sub>W<sub>x</sub>BC <u>Rizvi, Z.</u>;<sup>1,2</sup> Oliynyk, A.;<sup>1</sup> Tehrani, A. M.;<sup>1</sup> Brgoch, J.<sup>1</sup>

<sup>1</sup>Department of Chemistry, University of Houston, Houston, Texas <sup>2</sup>REU:REEMS Program, West Houston Center for Science & Engineering, Houston Community College, Houston, Texas

Superhard materials are widely used in industry as abrasives and coatings, for polishing materials, as mechanical parts, as well as an array of other industrial uses.<sup>1</sup> The current materials used for these applications involve precious metals, often osmium, rhenium, and iridium, or require extreme synthesis conditions to make like high pressure and high temperature.<sup>2,3</sup> This makes these materials costly and their production energy consuming. The development of new, alternative superhard materials that are not limited by resources, cost, or synthetic conditions has traditionally relied on searching for materials using traditional trial and-error-approaches. Recently, machine learning has shown promise as a tool for directing synthetic efforts towards materials with the most promising properties improving the chance of novel materials discovery. Unfortunately, hardness is a complex property and it is not currently possible to predict using machine learning directly. However, it is possible to machine learn a material's elastic constants, which are related to hardness, and are a fundamental for every material. This research employs machine learning to predict elastic constants and identify materials with high bulk and shear modulus to see if it indeed correlates with high hardness. The bulk and shear elastic moduli were predicted for ≈150,000 potential candidates that are compiled from crystallographic databases. From this large set of materials two compositions, ternary (ReWC<sub>0.8</sub>) and quaternary ( $Mo_{2-x}W_{x}BC$ ) compounds were selected considering they had the highest predicted bulk modulus among threecomponent and four-component inorganic compounds, respectively. ReWC<sub>0.8</sub> and Mo<sub>2-</sub> <sub>x</sub>W<sub>x</sub>BC compounds were synthesized using arc melting from stoichiometric mixture of the elements and their purity and crystal structure was confirmed using powder X-ray diffraction. The Vickers hardness was then measured using microindentation and demonstrated that ReWC<sub>0.8</sub> had a Vickers hardness of over 40 GPa (at 0.5 N applied load) whereas Mo<sub>0.9</sub>W<sub>1.1</sub>BC showed a Vickers hardness of 42 GPa (at 0.5 N applied load). The study proves that the novel machine learning approach to search for materials with desired mechanical property is viable to produce new superhard materials

*This work is supported by the American Chemical Society Petroleum Research Fund (55625-DNI10), the National Science Foundation Division of Civil, Mechanical &* 

Manufacturing Innovation (NSF CMMI 15-62142), and National Science Foundation Division of Materials Research (NSF DMR 14-60564)

References: [1] R. K. Bogdanov, A. A. Shulzhenko, A. P. Zakora, A. M. Isonkin, V. G. Gargin, New Superhard Material for Drilling Tools, *Journal of Superhard Materials*, **2007**, *29*, 56–63

[2] A. M. Terani, L. Ghadbeigi, J. Brgoch, T. D. Sparks, Balancing Mechanical Properties and Sustainability in the Search for Superhard Materials, *Integrating Materials and Manufacturing Innovation*, **2017**, *6*, 1–8

[3] H. Tracy Hall, Ultra-High-Pressure, High-Temperature Apparatus: the "Belt", *Review of Scientific Instruments*, **2004**, *31*, 125–131

![](_page_16_Picture_0.jpeg)

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**Kas** 

# oping ungsten-D **Mo<sub>2</sub>BC** of N Formed spu

Compour

### microscopy electron Y -N-B Ō diffraction experiment experiment experiment model Graphite impurity erimeı experim BC exp Mo BC W<sub>1,1</sub>Mo<sub>0.9</sub>BC W<sub>0.5</sub>Mo<sub>1.5</sub>BC BC 0.75 Mo<sub>1.25</sub>BC = Mo powder 2 ç Mo-W-B

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# Conclusions

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g method to finding method to finding to approach valuable learning σ <u>ທ</u> demonstrates that the novel machine with desired mechanical properties is properties perhard materials. produce new su

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study

This

# Work Future

and characterize machine learning method to synthesize compounds using superhard

synthesis methods to better control desired phase formation.

# Acknowledgements

Society are Foundation Research Experiences and Exploration in Shruti Hariyani (University Community We Chemical analysis. Science (Houston data American National Davis Is and DMR 14-60564) and Americarch Fund (55625-DNI10) for funding. and and experiments Gizzele Lim, the thank Amber Dr. i Viswanathan, Al for administering the NSF We and program. and DMR Sheinberg assistance Science Gayatri Bart for Petroleum Rese 15-6214 Houston) **Q** thank Materials College) grateful (CMMI We of

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H<sub>v</sub> =42.0

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![](_page_16_Figure_20.jpeg)

![](_page_16_Figure_21.jpeg)

![](_page_16_Figure_22.jpeg)

general formula W<sub>x</sub>Mo<sub>2-</sub> «BC sites, share two crystallographic Mo W and

applied load).

Vickers

The

v<sub>0.8</sub> has

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# Explore new

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Introduction:

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materials

Superhard

<u>Motivation</u>

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(H<sub>v</sub>) measurement

material as superhard.

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Vickers

and precious high pressure involve metals or require extreme synthesis conditions, like high temperature.<sup>1</sup>

superhard I trial-andtraditional ains unclear. potential σ The long term sustainability precious metals rem fo compared

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![](_page_16_Figure_34.jpeg)

B,G Х

Hardness

and compounds Ternary termetallic promising Quaternary compounds with highest predicted bulk moduli

predict

Simulations to

Machine Learning

Use

**Research objectives** 

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# diction

tic moduli 150,000 that are tic

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has compounds, W<sub>1.1</sub>Mo<sub>0.9</sub>BC hat the highest predicted bulk for highest for ternary quaternary compounds. Re<sub>0.5</sub>W<sub>0.5</sub>C has the predicted bulk for t

![](_page_16_Picture_40.jpeg)

elements. Purity and crystal structure was confirmed using powder X-ray diffraction. of the synthesized using arc melting from mixture Samples were

![](_page_16_Picture_42.jpeg)

![](_page_16_Picture_43.jpeg)

Re<sub>0.5</sub>W<sub>0.5</sub>C has a deficient carbon

![](_page_16_Figure_45.jpeg)

# Bulk Predicted Quaternar

[2] A. M. Terani, L. Ghadbeigi, J. Brgoch, T. D. Sparks, Balancing Mechanical Properties and Sustainability in the Search for Superhard Materials, Integrating Materials and Manufacturing Innovation, 2017, 6, 1–8

[1] R. K. Bogdanov, A. A. Shulzhenko, A. P. Zakora, A. M. Isonkin, V. G. Gargin, New Superhard Material for Drilling Tools, Journal of Superhard Materials, 2007, 29, 56–63

![](_page_16_Picture_49.jpeg)

# materials (Vickers Testing

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Vickers

![](_page_16_Figure_51.jpeg)

![](_page_16_Figure_52.jpeg)

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× strain В stress

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![](_page_16_Figure_60.jpeg)

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![](_page_16_Figure_63.jpeg)

![](_page_16_Figure_64.jpeg)

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#### Effects of Peptide functionalization on polyethylene glycol hydrogel and Biointeractions with oligodendrocyte progenitor cells Adel, Y.;<sup>1,2</sup> Nadeem, T.;<sup>1,2</sup> Lu, Xi.;<sup>1</sup> Smith Callahan, L<sup>1</sup>

<sup>1</sup>Vivian L Smith Department of Neurosurgery, Center for Stem Cell and Regenerative Medicine, McGovern Medical School, University of Texas Health Science Center at Houston, Houston, Texas

<sup>2</sup>REEMS REU program, West Houston Center for Science & Engineering, Houston Community College, Houston, Texas

Corresponding author:

Smith Callahan, L., Department or Neurosurgery, McGovern medical school University of Texas Health Science Center at Houston.

1825 Pressler Street, Institute of Molecular Medicine, Houston, TX 77030 Email: <a href="mailto:laura.a.smithcallahan@uth.tmc.edu">laura.a.smithcallahan@uth.tmc.edu</a>

Spinal cord injury (SCI) leads to neuronal and axonal injuries along with significant oligodendrocyte death. The loss of oligodendrocyte and subsequent demyelination contributes greatly to the pathological process. Oligodendrocyte precursor cells (OPC) are potential source of cell therapy as they can respond rapidly to injuries, proliferate quickly and differentiate into myelinating oligodendrocytes. We fabricated and tested a polyethylene glycol (PEG) hydrogel functionalized with different concentrations of extracellular matrix protein (laminin1 and 2) based peptides to see whether the hydrogel would serve as a suitable OPC delivery system. To measuring the mechanical material, hydrogels were loaded with different concentrations of each peptides (0-1000µM). No significant differences in Young's modulus, swelling ratio, water content and mesh size were observed between the different concentrations or type of peptides. Rat derived OPC were embedded in laminin1 peptide hydrogels and cultivated for three days. Pico green DNA measurements showed that while the amount of DNA increased from day 1 to day 3. However, there was no difference in the increase between different peptide concentrations. Differentiation capacity of the OPCs in gels were assessed via imaging and counting myelin basic protein (MBP) positive stained cells. The data showed that 250µM of laminin1 peptide trended towards higher percentage of MBP positive oligodendrocytes compared to other peptide concentrations. All concentration of laminin2 peptide showed similar percentage of MBP positive cells. These results showed that the current work with these peptide functionalized hydrogels serve as a foundation for a delivery system for OPCs which may be a suitable therapeutic for

restoring myelination and motor function after spinal cord injuries.

The authors gratefully acknowledge generous financial support provided by the National Science Foundation through NSF-DMR 1460564 and Ogilvie foundation.

![](_page_19_Picture_0.jpeg)

![](_page_19_Picture_1.jpeg)

![](_page_19_Picture_2.jpeg)

# Discussion

the 2 are known to affect OPC survival and function such as by enhancing myelin membrane formation. Peptides based on the functional domains of the proteins are easily tethered to the original proteins. We found that adding peptides to the hydrogels did not affect the major mechanical properties of the hydrogels such as Young's modulus, mesh size, and swelling ratios which suggests then that whatever biological effects that occur would then be due to the addition of the peptides and their concentration. Counting the MBP positive cells showed the percentage of differentiated r-OPC's was approximately 50-60% for both of the peptides. While 250μM concentration of laminin 1 trended higher percentage of differentiated OPCs no the cause the greatest levels of cell differentiation. Additional concentrations should be tested. It would also be interesting to see whether combinations of laminin 1 and 2 in the same gel and controllable base for developing cell delivery systems<sup>3</sup>. Proteins found in the extracellular matrix such as laminin 1 and concentration of each peptide may not have been optimized to flexible, easily modifiable were biological function of at adding peptides to due to the addition of the peptides that concentrations, there e reason may be One reason А the offers We found mimic compared to other laminin differences. PEG hydrogel affect cell function. may and significant lrogels Synthetic hyd will and

# Conclusion

cause significant  $\mathbf{C}$ Laminin Oſ peptides into 10% PEG hydrogel did not changes to mechanical properties of the gels Laminin of Embedding up to 1000µM peptides into 10% PEG hyc

and can survive, proliferate Within these hydrogels, OPCs

differentiate into oligodendrocytes Future studies will continue to optimize these gels for human OPCs and involve animal studies to test the viability of these system for restoring myelination and motor function after SCI.

# Acknowledgment

This work is supported by the HCC REEMS program through the National Science Foundation (DMR-1460564) and Ogilvie Foundation.

**1.** Silva, N. A., & Sousa, N. (2014). From basics to clinical: A comprehensive review on spinal cord injury. Elsevier: Progress in Neurobiology, 14, 25-57. Retrieved July 28, 2017.

., & Leung, G. K. (2015). Oligodendrocyte Precursor Cells in Spinal Cord Injury: A and Update [Abstract]. PMC4600489. doi:10.1155/2015/235195 **3.** Tsou, Yung Hao. "Hydrogel as a Bioactive Material to Regulate Stem Cell Fate." Bioactive Materials. New Jersey Institute of Technology, Sept. 2016. Web. 30 July Li, N. Review Fate.' 2017. Ci

# VCOL JC it gen Polvethylene **L**O Ť droc ON en uc Functionalizatic ligo ł Wil Interaction Peptide t S

Callahan<sup>2</sup> Smith *,aura* and  $Lu^2$ K lel<sup>1</sup>, Tasmia Nadeem<sup>1</sup>, Yasaman Ad

TX, USA. Houston center, **University of Texas Health Science** urgery, ; <sup>2</sup>Department of Neurosi **Community College** <sup>1</sup>Houston

# Results

![](_page_19_Figure_18.jpeg)

did the peptide pared to to aminin 2 **Fig 2:** Incorporation of Laminin 1 and Laminot significantly affect the Young's modulus control gel (without laminin 1/2). 500 oncentration [uM] 250 **Peptide Co**l

in 2	stdev	3.04	10.1	3.71	15.2	14.04	3.42
Lamin	average	17.5	30.2	16.6	20.4	30.7	16.1
in 1	stdev	10.54	1.323	2.04	8.61	10.27	14.25
Lamir	average	24.8	16.5	10.2	23.3	24.16	20.25
<b>Swelling Ratio</b>	Concentration	0	100	250	500	750	1000

6.1 17.2 7.7 31.02 30.8 9.2

2

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**Fig 3:**These tables indicate that there is no significant difference in mesh size, water content and swelling ratios between different concentration of peptide functionalized hydrogels. Fig 3 differ

stdev 1.12 1.52 1.6 3.5 2.1 1.5

nin 2

![](_page_19_Figure_22.jpeg)

![](_page_19_Figure_23.jpeg)

![](_page_19_Figure_25.jpeg)

antification of r-OPC differentiation in laminin 2 gels. itive for MBP is expressed as a percentage of the total f cells.  $\sim$ 300-400 total cells counted per gel were counted oncentration of peptide laminin 2 (n = 2 gels per group). **Fig 5:** Quantification of r-Cells positive for MBP is number of cells. ~300-400 for each concentration of pe

![](_page_19_Figure_27.jpeg)

**Fig 7**: R-OPC proliferation: Measuring DNA content with Pico-green of cells in functionalized Laminin 1(IKVAV) peptide hydrogel with a three day interval.

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100

80

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with the hydrogels

were differentiation gel size а 111

4% paraformaldehyde

![](_page_19_Picture_37.jpeg)

**Fig1:** Cells in peptide functionalized hydrogels stained with A2B5 indicate the cells are OPC's. Cells tend to grow in "oligospheres". A2B5(green)/DAPI(blue); Scale bar indicates 100 μm.

Mesh Size	Lamin	in 1	Lamin
Concentration	average	stdev	average
0	67.9	18.03	55.7
100	50.9	2.7	75.86
250	36.3	6.3	52.1
500	67.3	16.2	56.1
750	67.6	18.9	82.6
1000	57.6	29.9	52.5
Water Content	Lami	inin 1	Lami
Concentration	average	stdev	average
0	95.5	1.52	94.1
100	93.9	0.508	96.3
250	89.9	2.17	93.7
500	95.3	1.53	93.4
750	95.4	1.58	96.1
1000	93.01	4.74	93.6

![](_page_19_Figure_40.jpeg)

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# Effect

# Introduction

# Methods

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#### Effects of Peptide Functionalization on Polyethylene Glycol Hydrogel and Biointeractions with Oligodendrocyte Progenitor Cells.

<u>Tasmia Nadeem</u><sup>1,2</sup>, Yasaman Adel<sup>1,2</sup>, Xi Lu<sup>2</sup> and Laura Smith-Callahan.<sup>2</sup> Department of Neurosurgery, University of Texas, Houston, TX 77030

<sup>1</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College, Houston, Texas

<sup>2</sup>Department of Neurosurgery, University of Texas Health Science Center, Houston, Texas

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Callahan. L., Department of Neurosurgery, McGovern Medical School University of Texas Health Science Center at Houston.

1825 Pressler Street. Institute of Molecular Medicine. Houston, TX 77030

Email: Laura.a.smithcallahan@uth.tmc.edu

Spinal cord injury (SCI) leads to neuronal and axonal injuries along with significant oligodendrocyte death. The loss of oligodendrocyte and subsequent demyelination contributes greatly to the pathological process. Oligodendrocyte precursor cells (OPC) are potential source of cell therapy as they can respond rapidly to injuries, proliferate quickly and differentiate into myelinating oligodendrocytes. We fabricated and tested a polyethylene glycol (PEG) hydrogel functionalized with different concentrations of extracellular matrix protein (laminin 1 and 2) based peptides to see whether the hydrogel would serve as a suitable OPC delivery system. To measure the mechanical and material properties, ten percent PEG hydrogels were loaded with different concentrations of each peptides (0-1000µM). No significant differences in Young's modulus, swelling ratio, water content and mesh size were observed between the different concentrations or type of peptides. Rat derived OPC were embedded in laminin 1 peptide hydrogels and cultivated for three days. Pico-green DNA measurements showed that the amount of DNA increased from day 1 to day 3. However, there was no difference in the increase between different peptide concentrations. The differentiation capacity of the OPCs in hydrogels were assessed via fluorescence imaging and counting the number of cells positive for myelin basic protein (MBP) staining. The data showed that 250µM of laminin 1 peptide trended towards higher percentage of MBP expressing oligodendrocyte compared to other peptide concentrations. All concentration of laminin 2 peptides showed similar percentage of MBP positive stained cells. These results showed that the current work with these peptide functionalized hydrogels serve as a foundation for a delivery system for OPCs,

which may be a suitable therapeutic for restoring myelination and motor function after spinal cord injuries.

The author gratefully acknowledges support from Houston Community College REEMS REU Program and generous financial support from National Science Foundation Division of Material Research through NSF-DMR 1460564 and Ogilvie foundation.

![](_page_22_Picture_0.jpeg)

![](_page_22_Picture_1.jpeg)

![](_page_22_Picture_2.jpeg)

# Discussion

the 2 are known to affect OPC survival and function such as by enhancing myelin membrane formation. Peptides based on the functional domains of the proteins are easily tethered to the original proteins. We found that adding peptides to the hydrogels did not affect the major mechanical properties of the hydrogels such as Young's modulus, mesh size, and swelling ratios which suggests then that whatever biological effects that occur would then be due to the addition of the peptides and their concentration. Counting the MBP positive cells showed the percentage of differentiated r-OPC's was approximately 50-60% for both of the peptides. While 250μM concentration of laminin 1 trended higher percentage of differentiated OPCs no the cause the greatest levels of cell differentiation. Additional concentrations should be tested. It would also be interesting to see whether combinations of laminin 1 and 2 in the same gel and controllable base for developing cell delivery systems<sup>3</sup>. Proteins found in the extracellular matrix such as laminin 1 and concentration of each peptide may not have been optimized to flexible, easily modifiable were biological function of at adding peptides to due to the addition of the peptides that concentrations, there e reason may be One reason А the offers We found mimic compared to other laminin differences. PEG hydrogel affect cell function. may and significant lrogels Synthetic hyd will and

# Conclusion

cause significant  $\mathbf{C}$ Laminin Oſ peptides into 10% PEG hydrogel did not changes to mechanical properties of the gels Laminin of Embedding up to 1000µM peptides into 10% PEG hyc

and can survive, proliferate Within these hydrogels, OPCs

differentiate into oligodendrocytes Future studies will continue to optimize these gels for human OPCs and involve animal studies to test the viability of these system for restoring myelination and motor function after SCI.

# Acknowledgment

This work is supported by the HCC REEMS program through the National Science Foundation (DMR-1460564) and Ogilvie Foundation.

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., & Leung, G. K. (2015). Oligodendrocyte Precursor Cells in Spinal Cord Injury: A and Update [Abstract]. PMC4600489. doi:10.1155/2015/235195 **3.** Tsou, Yung Hao. "Hydrogel as a Bioactive Material to Regulate Stem Cell Fate." Bioactive Materials. New Jersey Institute of Technology, Sept. 2016. Web. 30 July Li, N. Review *Fate.*' 2017. Ci

# VCOL JC it gen Polvethvlene **L**O Ť droc ON en uc Functionalizatic ligo ł Wil Interaction Peptide t S

Callahan<sup>2</sup> Smith *,aura* and  $Lu^2$ K lel<sup>1</sup>, Tasmia Nadeem<sup>1</sup>, Yasaman Ad

TX, USA. Houston center, **University of Texas Health Science** urgery, ; <sup>2</sup>Department of Neurosi **Community College** <sup>1</sup>Houston

# Results

![](_page_22_Figure_18.jpeg)

did the peptide pared to to aminin 2 **Fig 2:** Incorporation of Laminin 1 and Laminot significantly affect the Young's modulus control gel (without laminin 1/2). 500 oncentration [uM] 250 **Peptide Co**l

in 2	stdev	3.04	10.1	3.71	15.2	14.04	3.42
Lamin	average	17.5	30.2	16.6	20.4	30.7	16.1
in 1	stdev	10.54	1.323	2.04	8.61	10.27	14.25
Lamir	average	24.8	16.5	10.2	23.3	24.16	20.25
<b>Swelling Ratio</b>	Concentration	0	100	250	500	750	1000

6.1 17.2 7.7 31.02 30.8 9.2

2

.⊆

**Fig 3:**These tables indicate that there is no significant difference in mesh size, water content and swelling ratios between different concentration of peptide functionalized hydrogels. Fig 3 differ

stdev 1.12 1.52 1.6 3.5 2.1 1.5

nin 2

![](_page_22_Figure_22.jpeg)

![](_page_22_Figure_23.jpeg)

![](_page_22_Figure_25.jpeg)

antification of r-OPC differentiation in laminin 2 gels. itive for MBP is expressed as a percentage of the total f cells.  $\sim$ 300-400 total cells counted per gel were counted oncentration of peptide laminin 2 (n = 2 gels per group). **Fig 5:** Quantification of r-Cells positive for MBP is number of cells. ~300-400 for each concentration of pe

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#### Andrew Catalanotto

### Solidification Front Velocity and Temperature Gradient in Metal Crystals Grown from the Melt

Andrew Catalanotto<sup>1</sup>, Daniel Suzuki<sup>2</sup>, Logan Ware<sup>2</sup>, Zachary Cordero<sup>2</sup>

<sup>1</sup> REEMS REU Program, The West Houston Center for Science & Engineering, Houston Community College, Houston TX

<sup>2</sup> Department of Material Science and Nano Engineering at Rice University, Houston TX

Directional solidification of single crystals and bicrystals requires a steep temperature gradient in the liquid phase to maintain a planar solid-liquid interface perpendicular to the growth direction. Knowledge of the thermal profile in the metal charge during growth can help guide the selection of optimal process variables (e.g., growth velocity). In this study, we directly measured the thermal profile during the directional solidification of a tin bicrystal using thermocouples sheathed in glass and placed in the center of the specimen. We then analyzed our measurements and determined the temperature gradient and solidification front velocity along the length of the specimen. Our experimental results help explain the physics of the crystal growth method under development, and confirm the benefit of incorporating the chill rod.

*This work is supported by the National Science Foundation (DMR-1460564) and the Houston Community College District.* 

# From S S S S S S Temperature Gradient in Metal Crysta Velocity and Solidification

Texas G. Ware<sup>2</sup>, Zachary C. Cordero<sup>2</sup> Logan uzuki<sup>2</sup>, SL Andrew M. Catalanotto<sup>1</sup>, Daniel H.

![](_page_24_Picture_2.jpeg)

![](_page_24_Picture_3.jpeg)

# Discussion

y 25% and produces a more
y 25% and produces a more and is uniform growth rate of the solid over time. Figure 4 shows a 2X increase in the temperature gradient in the liquid phase near the middle of the growth. Figure These results clearly demonstrate that the experimental configuration in Figure 2 can be used to analyze the effect of process variables on our modified 2 ° C/cm of the target value for the temperature gradient in tin [3]. increased gradient is seen throughout the growth, in Figure velocity by approximately growth rate of the solid over time. Figure the data Analysis of growth method. that this the

and increase in temperature gradient and reduction in growth velocity help explain Low temperature gradient and high growth velocity are factors that increase migration of grain boundaries in directionally solidified crystals [2]. Thus, the crystals, solidification, nucleation of stray the success of our modified Bridgman method with the chill rod. parallel of likelihood the

# **Future Works**

# ement analysis:

during profile measurements can take element to glass, and graphite are all very fragile and prone to failure. In order to rapidly reliably test a variety of geometries perform. The plaster, te are all very fragile used profile finite be can conditions, thermal software to days crystal growth. the growth simulate Thermal Finite el タ analysis to and and dn

![](_page_24_Picture_10.jpeg)

12

Temperature (°C)

![](_page_24_Picture_12.jpeg)

stal ethod. "helisoidal" Figure 6. Sinusoidal, helical, and "helisoidal geometries currently being explored using our Figure

# Testing complex geometries:

traditional birrin more and geometric geometrically complex crystal molds will variation affects the growth process. measurements of how Analysis understand so far profile σ analysis have only geometry. NS Thermal using help

25

of thermal profile using finite

# Acknowledgement

DMR 1460564 for this research opportunity. I would also like to thank Daniel H. Suzuki, Logan G. Ware and Dr. Zachary C. Cordero for their support and guidance during my research experience at The Additive Lab this summer. the National Science Foundation Division of Materials Research Grant

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Science for Houston Community College, Houston, TX d Nanoengineering, Rice University, Houston, <sup>1</sup> NSF Research Experience in Engineering and Materials Science and <sup>2</sup> Department of Materials

# Velocity Front ion

charge is plotted against the time at which the surrounding metal (tin) freezes. Figure 3 shows a ~25% decrease in growth velocity with the addition of the chill. By conducting latent heat away from the metal, the chill initiates freezing earlier in the growth and leads to a less steep, position of each thermocouple within the solidification front velocity, the more controlled growth velocity. calculate the Р

![](_page_24_Figure_24.jpeg)

Figure 3. Graph of the length solidified vs. time for a specimen both with (left) and without (right) a chill. Each point represents the time metal freezes around a thermocouple and its position along the length of the charge. The slope of the lines represents the growth velocity. dification Front Positio

# Gradient rature Temper

the is approximately double that with no chill throughout To calculate the temperature gradient (G), the temperature of the metal around each thermocouple is plotted against its position along the length of the charge each time the metal surrounding a thermocouple freezes. Figure 4 shows a plot during solidification, when the solid is ~5.1 cm long. Figure 5 shows the temperature gradient at each thermocouple as The graphs show that the solid. the length of it against gradient in the liquid ( $G_L$ ) with the chill solid is ~5.1 cm long. Figure solidification front passes it the growth. the

![](_page_24_Figure_28.jpeg)

Temperature Gradient at 5.1cm Without Chill

Figure 4. Graphs showing the thermal profile as the solidification front passes the center of the specimen both with (left) and without (right) a chill. Each point represents the temperature of a thermocouple when the solid is 5.1cm tall. The slope of the red lines represents the temperature gradient in the solid. Temperature Gradient Vs. Length of Solid Gradient in Solid -Gradient in Solid

![](_page_24_Figure_31.jpeg)

![](_page_24_Figure_32.jpeg)

l thank

Figure 5. Graph showing the temperature gradient as the solidification front passes each thermocouple in the specimen with a chill(left) and without a chill(right). The dotted black line represents the target gradient in the liquid phase. Each point represents the temperature gradient as the solidification front passes a different thermocouple within the charge. Length of Solid (cm) Gradient in Liquid ent in Solid Length of Solid (cm) nt in Liquid Gradie ent in Solid -Gradie

10

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2

0

decreases Bridgman shows within ~ Figure 7. Simulation element analysis.

# Method Bridgman The Š Solidification Directional

remove Various Directional solidification is the process of cooling metal castings by freezing the end. eliminate cavities, impurities from metals, and grow large single crystals and bicrystals. casting to the opposite to used are end of the solidification metal progressively from one of directional methods

ch on crystalline Figure 1 depicts on used in this gradient in the As the at the one then the stray for thod is the growth of zone rises above the base of SO method of directional solidification [1]. The primary application of the is completely in the apply a furnace continues to rise. Applying the liquid phase and maintaining a ത velocity interface, uration used in t furnace is lowered and bicrystals along as preventing the formation of <u>.</u> liquid results freezing begins of the mold. then raised to front top method gradient liquid temperature the solidification research configuration the velocity) Bridgman to crystals microstructure solid that the mold Bridgman met of temperature vertical axis The mold, progresses zone, scientific steep (growth bottom planar single study. slow The hot hot the the

In this work we fabricate molds capable of measuring the thermal profile during conditions are in the growth, then analyze the profile to explain why our growth crystals [2].

# imen

1.275 cm probing of the metal Fabrication of specialized molds to measure the thermal profile during growth is **Thermal Measurement Molds** necessary due to the mold geometry, which prevents direct probing of the along its axis as the crystal is grown. Glass tubing is invested in the plaster, 50 10 Plaster mold Steel flask Tin charge Glass tube Chill rod-The to poured plaster, the without experimental are temperature (500°C). allows the measurement of internal failure due to molten metal leaking out of the mold. Once the metal inserted into the glass tubes. The charge is then directionally solidified glass tubes in plaster and increasing pattern, from thermocouples old. Once the "charge" - is <u>.</u> mold shield the thermocouples the with the positive of temperature of the damaging the mold, through probability mold, the Investing the growth known as the drilling of along high into the

![](_page_24_Figure_43.jpeg)

Figure 2. Cross section diagram of molds used to measure the thermal profile during crystal growth with a chill rod (left) and without a chill rod(right).

while

Bridgman method

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using

its

along

recorded

<u>.</u>

data

thermal

length.

# Solidificat

![](_page_24_Figure_46.jpeg)

![](_page_24_Figure_47.jpeg)

![](_page_24_Figure_48.jpeg)

![](_page_24_Figure_49.jpeg)

![](_page_24_Figure_50.jpeg)

![](_page_24_Figure_51.jpeg)

![](_page_24_Figure_52.jpeg)

![](_page_24_Picture_54.jpeg)

![](_page_24_Picture_55.jpeg)

# **Modified Bridgman Method**

![](_page_24_Picture_57.jpeg)

Figure 1. Cross section diagram of modified Bridgman setup with moving furnace directionally solidifying a stationary metal casting. The graphite chill is inserted into the steel pipe at room temperature to act as a heat sink, increasing the temperature gradient along the vertical axis.

# optimal range.

Spec Measurement Thermal

#### **Nathaniel Ocanas**

#### Poster Number Five

#### **Fabrication of Bulk Nanocrystalline Alloys Using Ultrasonic Powder Compaction** Nathaniel Ocanas<sup>1</sup>, Raquel Torres<sup>3</sup>, Austin Ward<sup>2</sup>, Zachary Cordero<sup>2</sup>

<sup>1</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College, Houston TX

<sup>2</sup>Department of Material Science and Nano Engineering at Rice University, Houston TX <sup>3</sup>NSF Research Experience for Teachers, Rice University, Houston, Texas

Nanocrystalline alloys have high strengths, high hardness values, and excellent wear resistance, making them attractive materials for use in numerous structural applications. However, it remains a challenge to reliably fabricate bulk nanocrystalline materials because the most straightforward synthesis method, powder consolidation, involves the densification of nanostructured powder at high temperatures. This approach presents challenges because nanocrystalline materials exhibit rapid grain growth at elevated temperatures, resulting in the loss of their unique properties, attributed to fine grain sizes. One low-temperature method that could be used to consolidate nanocrystalline powders without significant coarsening is ultrasonic powder compaction. In this process, a punch compresses feedstock powder in a die and then shears the powder at ultrasonic frequencies transverse to the loading axis. This combination of normal pressure and oscillating shear stresses disrupts the native oxide on the feedstock powder particles, resulting in metallurgical bonding. Although elevated temperatures are not required for ultrasonic powder compaction, they can result from frictional heating. Therefore, to optimize the grain size and mechanical properties of our compacts, we have identified the set of ultrasonic powder compaction variables (e.g., uniaxial compaction pressure, compaction time, oscillation amplitude) that will cause minimal heating, but still produce a fully dense compact. Ultimately, this work will inspire future work on the fabrication of bulk nanocrystalline materials and improve our understanding of the mechanical properties of nanocrystalline materials.

This work is supported by the National Science Foundation (DMR-1460564), the Houston Community College District, and Rice University.

# er Compaction

![](_page_26_Picture_1.jpeg)

Ο

![](_page_26_Picture_2.jpeg)

# **Conclusions & Future Work**

![](_page_26_Figure_5.jpeg)

the be 9 diffraction × measurements to from figure size predicted ~30 nm. results grain ray 

> results from Figure 6. X-ray Diffraction results densified nanocrystalline Copper

ultrasonic densification and the mechanical properties of the powder nanocrystalline effect the clarify uo have findings vibrations compacts. These

- of with densification scales density ultrasonic oscillation amplitude aid the Ultrasonic vibrations Compact powders.
- decreases with increasing compacts Hall-Petch The these pressure. retained their nanostructure. that Compact hardness predicts uniaxial equation applied  $\bullet$

of Next, we will explore the effect of the process structure parameters and temperature on the the feedstock powder. We will:

process parameters to the temperature change analytical model which relates the thus the final structure in the compact an Create

preserve compact nanostructure during densification. to NS enable will next steps These

# Acknowledgements

of research Dr. Torres for their support and ту Division Austin A. Ward, as well e Foundation Divis 1460564 for this r development as both a researcher and engineer. as I thank the National Science Foundation Materials Research Grant DMR 1460564 for t opportunity. I would also like to thank Austin Zachary Cordero and Raquel M. Torres for their mentorship during my research experience as

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  [7] Cordero, Z. C., B. E. Knight, and C. A. Schuh. International Materials Reviews 61.8 (2016): 495-512.

Ward <sup>2</sup> , Raquel M. Torres <sup>3</sup> , Zachary C.	Material Science for Houston Community College, Houst noengineering, Rice University, Houston, Texas e for Teachers, Rice University, Houston, Texas	rimental Details	milled 99.9% / ball mill for 5 rocess control	oscillation direction sonotrode stock powder	at ultrasonic tion. Ind oscillating the feedstock gical bonding it within close Figure 1. Schematic of the ultrasonic powder consolidation apparatus using an instrumented ultrasonic welder.	ry & Modelling	mpact Eq 2, the Hall-Petch equation, relates the yield ler an stress of polycrystalline materials to their grain size Eq 1 (D). In this work, we use Eq 2 to estimate the grain f our size of our compacts by measuring their hardness, which is $3\sigma_Y$ , and using material constants $\sigma_0 =$ 40 MPa and k = 110 MPa · $\mu m^{1/2}$ found in literature [7]. $\sigma_Y = \sigma_0 + \frac{k}{\sqrt{D}}$ (2)	<b>Densification Results</b>	ensity Figure 5 shows experimental hardness d at measurements and grain size predictions by Eq 2. es up We found higher pressures put out compacts with lower hardness values.	itude : Paradroess (HV) Paradroess (HV	itude : hum itude : 100 100 100 100 100 100 100 10	Copper Figure 4. Nanocrystalline Copper Figure 5. Hardness as a function of creasing micrographs of Vickers Hardness pressure
Allo A. Ward	nd Materi Vanoengir nce for Tea	erimen	ve milled ergy ball mi process o	<b>Process</b> eedstock p	at ultr rection. e and osci on the fee lurgical bo ught withir	eory & I	compact under an er, Eq 1 for the for the ion.	r Densif	density med at cales up	amplitude : 13 µm	amplitude : 15 µm	line Copper I increasing r

# Nathaniel L. Ocañas<sup>1</sup>, Austin A. of Bulk Nanocrystalline

<sup>1</sup> NSF Research Experience in Engineering and

<sup>2</sup> Materials Science and Nan

Experience <sup>3</sup> NSF Research

# Experi

a pro SPEX high energy We as powder, added feedstock with 0.8% stearic acid metals basis Cu powder in a our synthesize hours agent. 2

A punch (sonotrode) compresses the feeds **Ultrasonic Powder Compaction Pro** 

- in a die. Н.
  - σ frequencies transverse to the loading direct powder the shears sonotrode The Ч.
- an shear stresses disrupts the native oxide on t when the particles deform and are brought powder particles, resulting in metallurg pressure normal of combination contact. The  $\tilde{\mathbf{\omega}}$

# Theor

# **Powder Densification**

of

unde density (*p*) of a powder com for of ultrasonic vibrations which aid in densification. However, density account with yield stress  $\sigma_{Y}$  due to plastic yielding applied uniaxial pressure ( $\sigma$ ) [6]. Howev the not dramatically underestimates it does compacts because gives the μ Eq

## (1) $\left(\frac{-3\sigma}{2\sigma_{Y}}\right)$ exp ( — Q

# Powder D

del scales performed Generally, density experimental compactions with increasing amplitude shows different amplitudes. for measurements N Figure

of

to

![](_page_26_Figure_42.jpeg)

Figure 3. Nanocrystalline C Micrographs Amplitude Figure 2. Density as a function of peakpeak amplitude

of

![](_page_26_Figure_44.jpeg)

0.86

0.88

![](_page_26_Picture_45.jpeg)

![](_page_26_Picture_46.jpeg)

![](_page_26_Picture_47.jpeg)

# Nanocrystalline Materials

their small grain size Nanocrystalline materials have many uniq (< 100 nm). For example: properties because of

ue

- Advantages
- High strength [1]
- High hardness [2]
- Strong magnetic properties [3]
  - Excellent wear resistance [4]

- Good thermoelectric properties [5]
  - Disadvantages
- Unstable at high temperatures Difficult to synthesize in bulk

# **Motivation for Project**

- them industries aerospace, among other structural applications. make properties automotive for applications in the materials superior and nanocrystalline biomedical, ideal The T
- of bulk nanocrystalline specimens would enable scientific work nanocrystalline materials which is currently impossible. of behavior The fabrication the UO 2.

# Solid State Bonding

sly ulk synthesized nanostructured material into b synthesizing large quantities previou attempt of We amounts materials, small nanocrystalline Rather than components. consolidate

desired powder once bonding We Because nanocrystalline materials can not be temperature arc welding, to densify bulk material σ ultrasonic nanocrystalline powder is milled to solid-state welding techniques, such as arcexamine use We particle size. compaction techniques. examine joined

#### **Frank Kornet**

#### Poster Number Six

#### Investigating Portability of Molecular Dynamics Applications <sup>(1)</sup> Frank Kornet <sup>(2)</sup>, Millad Ghane <sup>(3)</sup>, Pengzhi Zhang <sup>(3)</sup> and Margaret S. Cheung <sup>(3)</sup>

Most scientific applications will have to be revisited to run on Graphic Processor Units (GPUs) to obtain speedup with modern architectures. Compute Unified Device Architecture (CUDA) is widely used within the scientific community as the proprietary NVIDIA GPU programming standard. OpenACC is the open, standard alternative. A careful comparison between the two is lacking in terms of productivity, portability and performance.

CoMD, a proxy for molecular dynamics applications developed by national laboratories, is exploited to assess abovementioned criteria. A partial OpenACC and a full CUDA version of CoMD exists that try to investigate capabilities of each programming model. The partial OpenACC version is modified by porting one of the remaining functions to OpenACC. We evaluated the performance of both implementations on a single computer node with two NVIDIA K80 GPUs.

Experiences with the porting show that OpenACC is learned in about two weeks. The lines of code show that OpenACC is more productive than CUDA (3,704 vs. 6,781) and in terms of portability, CoMD runs successfully on multiple architectures. After applying different strategies on the partial OpenACC code to improve performance, OpenACC speedup over serial CoMD reaches 7.2x while CUDA speedup still dominates (61.1x). Investigations show that changes have been made to the underlying CUDA CoMD algorithm. A firm conclusion regarding performance cannot be drawn at this stage as the two algorithms are different.

<sup>1</sup>Work funded by National Science Foundation grants PHY-1427654, ACI-1531814, MCB-14122532, and DMR-1460564.

<sup>2</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College

<sup>3</sup>University of Houston

![](_page_28_Picture_0.jpeg)

![](_page_28_Picture_1.jpeg)

![](_page_28_Picture_2.jpeg)

![](_page_28_Picture_3.jpeg)

# Productivity and Ease

- OpenACC needs fewer lines of code than 1-2 weeks. Once learned, developers can start to optimize their applications. OpenACC basics can be learned in
- CUDA, but different algorithm overstates CUDA part

	Serial	OpenACC	CUDA
Lines of Code	3,407	3,704	6,781

# Portability

- UO successfully tested and run NVIDIA Quadro 600 and K80 CoMD was
  - aS Non-NVIDIA GPUs were not tested part of this study

![](_page_28_Figure_12.jpeg)

# Acknowledgement

Frank Kornet would like to thank Millad Ghane, Pengzhi Zhang, Bart Sheinberg, Sunita Chandrasekaran and Margaret S. Cheung for providing him the opportunity and helping him along the way. Frank would also like to thank National Science Foundation (NSF) for providing funding for the project.

![](_page_28_Picture_15.jpeg)

# Portability of Molecular Dynamics Applicati 50 C

Frank Kornet<sup>(1)</sup>, Millad Ghane, Pengzhi Zhang and Margaret S. Cheung

# Objectives

and more productive than CUDA

20 per cent of CUDA's performance within 10

# Proxy as CoMD Use

simulation (NVE) and is developed by DoE Co-design Center for Materials in Extreme Environments (ExMatEx). dynamics (MD) is the method of simulating the kinetic and thermodynamic properties of molecular f this class. It carries out a microcanonical ensemble systems using Newton's equations of motion and represents an important class of scientific applications. as it is both small and representative o CoMD is used Molecular

![](_page_28_Picture_23.jpeg)

Energy + Kinetic Energy

study. Please note that CoMD results provide an upper bound of what is achievable with algorithm. For large simulations, CoMD will split the system and distribute available on GitHub and a full CUDA version are cells over participating nodes. A partial OpenACC version CoMD implements an O(N) Verlet  $O(N^2)).$  $\mathbf{\uparrow}$ GPUs in practice (O(N) are used in the and

dded Atom Model (EAM) geoevo\_sfc\_osa (3) Embe nt?utm\_campaign=shopping\_ g/288728381/real

Results

![](_page_28_Figure_27.jpeg)

|X|X|X|X

(\*)

computeForce ( advanceVelocity kineticEnergy

- portable than CUDA use that OpenACC is easier to is more OpenACC that Demonstrate Demonstrate
  - OpenACC's performance is that Demonstrate

(2): https: s/f/fac bsite/pa (1):

GPU X |×|× CPU  $\mathbf{X}$  $\mathbf{X}$ ade updateLinkCells haloExchange sortAtomsInCell uteForce (\*) Σ printThings advanceVelocity advancePositions redistributeAtoms Change sumAtoms redistri timestep Function main

![](_page_28_Figure_32.jpeg)

(1) Email Frank Kornet: W209007857@student.hccs.edu

 Synthesis of Magnesian –Iron Silicate Phases by the Solid State Method Arceneaux, B. <sup>1,2,</sup>, Martinez, M. <sup>1,2,</sup>, Vu, B.<sup>1,2,</sup>, Meen, J.<sup>2,</sup>, Müller, K.<sup>2,</sup>
 <sup>1</sup>REEMS REU Program, Houston Community College, Houston, Texas 77082
 <sup>2</sup>Texas Center for Superconductivity, University of Houston, Houston, Texas 77204

Corresponding Author: James K. Meen, Department of Chemistry and Texas Center for Superconductivity, University of Houston 3369 Cullen Blvd Rm 202, Houston, Texas 77204. E-mail: jmeen@uh.edu

Many magnesiun silicates have extensive solid solutions with iron substituted for magnesium and properties of the solutions vary with Mg:Fe. We have synthesized bulk compositions that should crystalize (Mg,Fe)<sub>2</sub>SiO<sub>4</sub> (olivine) and (Mg,Fe)SiO<sub>3</sub> (pyroxene) by grinding together suitable pure powders (MgCO<sub>3</sub>, SiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>) and sintering them. The sintered pellet was broken and reground three times and then pressed onto a platinum loop which was held in a vertical tube furnace at temperatures between 1300 and 1400 °C. Experimental changes were first studied by scanning electron microscope to determine phase assemblages and then, after polishing, by electron microprobe analyzer to determine precise atomic compositions of phases. Results of selected experiments will be shown and principles of the kinetics of these systems discussed.

This work is supported by REEMS (NSF DMR 1460564) and by funding from the State of Texas to JKM through the Texas Center for Superconductivity.

![](_page_30_Picture_0.jpeg)

2

# Method **Solid State** Ð

77082

ston, Texas 77204

# CONCLUSION

Synthetic yielded of being to the air the incorporated into the silicates, probably because the iron remained in the trivalent state (Fe<sup>3+</sup>). Synthetic synthesize component Comparison of the electron micrographs in in initial experiments the iron, reduces to the stituted for Mg<sup>2+</sup> i not environments oxides their to performed substituted from iron reducing show that with silicates homogeneous material as  $(Fe^{2+})$ heterogeneous were the run products in magnesium-iron state Experiments experiments silicates. divalent oxides. were

# FUTURE WORKS

Energy constant energy differences between the Substitution of a rare element on the octahedral site of the silicate can produce a chroma center if the rare earth can absorbed in any form to excite the rare earth will be re state. of SO occupy excited states as well as a ground and quantum wavelength reflecting the fixed а as two states. released

# FUTURE USES

to for exploration. Sustainability is a concern for the space mission. This process to make glow sticks to send a space shuttle initiative to be used as a form of light. an has NASA MARS

# REFERENCES

- 1. James K. Meen, pers.comm.
- of a Mixture." retrieved from https://www.Quia.com/files/quia/users/ Components "Separating the сi
- and Homogeneous Mixtures?" ThoughtCo., retrieved from "What Is the Difference Between Heterogeneous https://www.thoughtco.com/heterogeneous-andhomogeneous-mixtures-606106  $\tilde{\mathbf{C}}$

# •ACKNOWLEDGEMENT

•This work is supported by REEMS (NSF DMR 1460564)

•And in part by Grants from the State of Texas to JKM through the Texas Center for Superconductivity.

![](_page_30_Picture_18.jpeg)

# Magnesiu of Synthesis

1. RE Arcen Tex Ч.

# **PROCESSES**

# Separating the Components of a Mixture

- separa worl а g g of different use substances that water encounter them on the We It is not difficult to ordinary in fuel materials advantage the soil, air, and e common mate more consume, chemistry; we es. In fact, very few common Any material made up of two or if you take physical properties of the components. chemically combined is a mixture. We materials, drinks components of a mixture to Mixtures are not unique basis. The food and c basis. The food and vehicles, building mai fact, In mixture pure.
- OTC easily, leaving beh matter with high vaporization temperatures, whether liquid, a mixture in Distillation. This is the process of heating that evaporates more liquid а remove or both.
- Sublimation. Some substances change directly from a solid to when the vapor caffeine back to the solid phase without a liquid state in between, is Dry ice sublimes also. iodine, are The reverse process, sublime paradichlorobenzene (mothballs). Some solids which when they are heated. deposition.

the

- S dissolve А With this technique, substance can be separated from an insoluble substance. selectively to solvent solid mixture. А This uses component of the Extraction.
- This separates a liquid from insoluble solid sed by carefully pouring the liquid from the solid without disturbin Decantation. solid.
  - This separates a solid from a liquid through the us These materials trap the solid but let the liquid pass thro Paper, charcoal, or sand can serve as a filter. porous material Filtration. filter.
- To separate the components of a mixture using physical meth-Purpose: •
  - To calculate the percent composition of a mixture. 2

Neon

Argon Rejactary Krypton Krypton

4.002602 23723 Helium

![](_page_30_Figure_31.jpeg)

# UNIVERSITY of HOUSTON

TEXAS CENTER FOR SUPERCONDUCTIVITY

![](_page_30_Picture_34.jpeg)

WHERE DISCOVERIES BEGIN

# INTRODUCTION

# ABSTRACT

1300 solid and SiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>) and sintering them. The sintered pellet was broken and reground three to sintered pellet scanning electron microscope to determine phase We have (Mg,Fe) $_2$ SiO $_4$  (olivine) and (Mg,Fe)SiO $_3$  (pyroxene) by . Experimental changes were first studied atomic selected crystalize electron of magnesium a platinum loop which was held at temperatures between extensive principles precise assemblages and then, after polishing, by should of properties of the solutions vary with Mg:Fe. synthesized bulk compositions that should Results determine substituted for have kinetics of these systems discussed. and shown silicates phases. to will be analyzer vertical tube furnace iron magnesian of with onto compositions and 1400 °C experiments microprobe solutions pressed Many þ

σ

# TABLE PERIODIC

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period 1	2	3	4	5	9	7	

131. Xenon wolds<sup>426</sup> 65<sup>6</sup> 66<sup>6</sup>

Radon [Xe] 4\*\* 5d\* 66\*

|₽

#### Partition of the Phases in a Ternary System Phosphor and the Effect of the Crystal Structures Created on Luminescence

Martinez, M.<sup>1, 2</sup>; Arceneaux, B.<sup>1, 2</sup>; Vu, B.<sup>1, 2</sup>; Meen, J.K.<sup>2</sup>; Müeller, K.<sup>2</sup>

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<sup>2</sup>Texas Center for Superconductivity, University of Houston, Houston, TX 77204.

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When a solid is sourced with a form of energy, it may emit photons in excess of thermal radiation in a process called luminescence. In inorganic solids, luminescence is emitted as the result of electronic transitions between quantum mechanical states with different levels of energy. Transitions through intermediate metastable states determine the duration of the luminescence, if prolonged after the supply of energy is cut off from the solid it is referred to as phosphorescence. Phosphors are used as powders in fluorescent lamps, cathodoluminescent television screens, and are a promising future source for alternative lighting (Yacobi & Holt)<sup>[1]</sup>. However, it is unclear how the phases present in a ternary system phosphor affect its luminescence. To tackle this, the ratio of concentrations in the compound of the mixture of immiscible phases were explored in a series of powdered Mg<sub>2</sub>SiO<sub>4</sub> samples doped with  $Mn^{2+}$ , and co-activated with an Al<sub>2</sub>O<sub>3</sub> impurity and rare earths (such as  $Dy^{3+}$  and  $Eu^{3+}$ ), which were prepared and sintered with 95% Ar, 5% H<sub>2</sub> gas at different temperatures in a tube furnace, and treated with different quenching techniques in a thermocouple system, in order to achieve cathodoluminescence, and homogeneity in the system of samples. A Scanning Electron Microscope (SEM) was used to look for both mentioned characteristics above, and Electron Microprobe Analysis (EMPA) was employed to obtain the precise chemical compositions in the crystal structures of the compounds. Samples that were coactivated by Al<sub>2</sub>O<sub>3</sub> displayed cathodoluminescence more intensely, as opposed to those that were not. These samples, in particular were observed to have spinel, corundum, and a clear solid state cordierite crystal structure exhibiting the luminescence phenomena. By investigating the effects of the partition coefficient in this specific phosphor it creates the opportunity of formulating a more effective luminescent outcome for future applications.

This work is supported by the National Science Foundation Division of Materials Research REEMS (NSF DMR 1460564) and by grants from the State of Texas to JKM through the Texas Center for Superconductivity.

<sup>[1]</sup> Yacobi, B. G., and D. B. Holt. *Cathodoluminescence Microscopy of Inorganic Solids*. New York and London: Plenum Press, 2010. Print.

![](_page_33_Picture_0.jpeg)

![](_page_33_Picture_1.jpeg)

**S** 

National Science Foundation

# ussion

ase relations and phase compatibility relations in MnO,  $_{2}$ O, SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub> were investigated in solid state environment. oeriments in a reducing

freezing point depression, although the crystallization Al<sub>2</sub>O<sub>3</sub> caused a temperatures of both spinel and cordierite were Addition of MnO to the MgO,  $SiO_2$ , supressed.

The multiply saturated liquid line (spline & cordierite) moved away from the spinel composition point. Partitioning of Mn between the phases was consistent with the phase relations

Pure Mg spinels and cordierite are white and do not luminesce, in the spectral range investigated.

Mn bearing spinels exhibited significant cathodolumnescence. Coexisting Mn bearing cordierite did not exhibit the cathodoluminescence

# Ire Work

- Calculation of Gibbs Free Energy
- **Check for Photoluminescence**
- Measure the Wavelength of Luminescence
- Add Rare Earths

# owledgements

and This work is supported by the National Science Foundation Division of Materials Research REEMS (NSF DMR 1460564) by grants from the State of Texas to JKM through the Texas Center for Superconductivity. Special thanks to Dr James K. Meen and Dr Karoline Mueller at the Texas Center for Superconductivity.

# rences

rganic Solids. New York and London: Plenum Press, 2010. Print rvey, David. "10.6: Photoluminescence Spectroscopy." *Chemistry reTexts*. Libretexts, 24 Dec. 2016. Web. cobi, B. G., and D. B. Holt. Cathodoluminescence Microscopy of

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SON LUMINESCEN	cer for Superconductivity, University of	tructures Created	oscope (SEM) Analysis cence & Homogeneity	alysis (EMPA) Concentration of Compounds	1       1	of Mn in Crystal Structures	$\frac{inel}{lass} = 0.37$	$\frac{rdierite}{Glass} = 0.14$	$\frac{Spinel}{rdierite} = 2.67$	EFFICIENT VS. TEMPERATURE iorite — Linear (Spinel) — Linear (Cordiorite)		1340         1350         1360         1370         1380         1390           TEMPERATURE [°C]         1360         1370         1380         1390		$\begin{tabular}{c} & & & & & & & & & & & & & & & & & & &$	of Black lines illustrate the interrelations of phase fields in pure MgO, SiO <sub>2</sub> , Al <sub>2</sub> O <sub>3</sub> (MAS) ternary system, and the blue ones determine the shift caused by Mn in the MAS system.

### M Ш Ш Vu, Ŭ 1, T Ìı M Ш Т Arceneaux Ŋ Ш 7 1, 2. Ш N. Ľ 0 L Ш Ш

<sup>2</sup>Texas

![](_page_33_Figure_24.jpeg)

#### **Brandon Vu**

#### **Experimenting With Cobalt Doped Phosphors and Phosphorescence**

Brandon Vu<sup>1,2</sup>, Monica Martinez<sup>1,2</sup>, Brenda Arceneaux<sup>1</sup>, and Dr. James Meen<sup>2</sup>

<sup>1</sup>REEMS REU Program, the West Houston Center for Science & Engineering, Houston Community College, Houston TX, 77082 <sup>2</sup>Texas Center for Superconductivity, University of Houston, Houston TX, 77024

Corresponding Author: James K. Meen, Department of Chemistry and Texas Center for Superconductivity, University of Houston 3369 Cullen Blvd. Rm 202, Houston, TX 77024 (Email:jmeen@uh.edu)

As nations around the globe continue to reduce fossil fuel usage, a renewable resource must take its place. Phosphorescent materials have recently begun to show their potential as a new source for renewable light. Currently, Mn doped phosphors mixed with rare earths have shown emission times of 10 hours. However, red light emission only last four hours. To combat this problem we created a Al-Si-Mg sample doped with Co, another transition metal, which will hopefully provide a basis for a new phosphorescent material when mixed with a rare earth. The sample was created by mixing  $Al_2O_3$ ,  $SiO_2$ , MgCO<sub>3</sub>, and Co<sub>2</sub>O<sub>4</sub> powder and compressing the sample into a platinum loop. The sample was then heated to 1350 °C for 10 minutes with forming gas  $(95\% \text{ Ar}-5\% \text{ H}_2)$  before reducing the heat and leaving it for 30 minutes. The lower equilibration temperature was varied over multiple trials and ranged from 1150-1370 °C to create the crystal phases cordierite, forsterite, and spinel. The finished samples were examined under a scanning electron microscope (SEM) and showed signs that the three phases exist within the samples. Further examination under a microprobe will determine precise chemical compositions that will allow determination of elemental partition coefficients between phases and provide details on the influences of the contents of cobalt and rare earth elements (Dy, Eu, etc.) on the luminescence, if any, of these silicate phases.

This work is supported in part by the National Science Foundation Division of Materials Research REEMS (NSF DMR 1460564) and in part by grants from the state of Texas to JKM through the Texas Center for Superconductivity.

escence	Dline Mueller <sup>1</sup> , Texas, 77024	

# Conclusion

signs of any luminescent properties from any sample

0 Z

Future research could analyze the new phase addition of Co, Eu, and Tb rather than a shift Data indicates new phase relations with the relations

Creation of new phase diagram

![](_page_35_Figure_6.jpeg)

Temperature

# Acknowledgements

Special thanks to Dr. James Meen and Dr. Karoline Mueller at the Texas Center for Superconductivity

Foundation (DMR 1460564) and grants from the This work is supported by the National Science state of Texas to JKM at the Texas Center for Superconductivity

# References

Yellow, and Red Long Persistent Phosphors Xiao-Jun, W., Dongdong, J., 2003, Mn Activated Green,

# ped Phosphors and Phosphor 0

1.Department of Chemistry and Texas Center for Superconductivity, University of Houston, 3369 Cullen Blvd, Houston , Dr. Karc 2. REEMS Program, Houston Community College, 2811 Hayes Rd., Houston, Texas 77082 James Meen<sup>1</sup> Dr. Arceneaux<sup>1,2</sup>

- oxide
- and add 2.5% europium and 2.5%

 $\bigcirc$ 

# ults

# Crystal Phases Observed (Sample 2)

- Spinel
- Enstatite
  - Olivine
- Observations
- Cobalt seems to have completely replaced Aluminum in spinel
  - Results in new phase relations  $\bullet$

# $\widehat{\mathbf{m}}$ 2 new phases that haven't been **Crystal Phases Observed (Sample** properly characterized $\bullet$

- $Mg_2Al_2Si_2O_9$ lacksquare

  - $Mg_2Al_2Si_2O_{15}$

![](_page_35_Figure_32.jpeg)

### Effect of Side Chain Length on the Physical Properties of Linear Polymers with a Transient Network

Scholtes, Kevin<sup>1,2</sup>; Ding, Wenyue<sup>1</sup>; Robertson, Megan.L<sup>1</sup>

<sup>1</sup>Department of Chemical and Biomolecular Engineering, University of Houston, Houston, Texas <sup>2</sup>REEMS Program, West Houston Center for Science & Engineering, Houston Community College, Houston, Texas

Corresponding author:

Robertson M. L., Department of Chemical and Biomolecular Engineering, University of Houston 4726 Calhoun Road, S222 Engineering Building 1, Houston, TX 77204 E-mail: <u>mlrobertson@uh.edu</u>

The increasing demand to create innovative and sustainable materials with improved quality while reducing overall costs is crucial in order to overcome the challenges of scarcity of raw material sources such as petroleum oils. An attractive solution is to create biorenewable polymers with desirable properties that are derived from renewable resources. Vegetable oils and their fatty acids are useful raw material sources for polymers due to their lack of toxicity, worldwide availability, and ease of functionalization. We probed the properties of copolymers derived from fatty acids, poly(stearyl acrylate-co-lauryl acrylate). The two acrylate monomers differ in the length of their side chain, in which stearyl acrylate and lauryl acrylate contain 18 and 12 carbon atoms per side chain, respectively. In order to improve their physical properties, we have incorporated a transient network in the copolymer through copolymerization with acrylamide, which undergoes hydrogen bonding. We examined the effect of the side chain length of the polyacrylate on physical polymer properties, including melting temperature and rheological properties, both with and without copolymerization with acrylamide. The melting point increased with increasing stearyl acrylate content in the copolymer. Copolymerization with acrylamide had negligible effect on the melting temperature. Copolymers containing acrylamide exhibited higher storage and loss moduli. The presence of hydrogen bonding produced deviations from typical terminal region behavior.

This work is supported by the National Science Foundation (CMMI-1334838 and DMR- 1611376), the HCC REEMS Program (National Science Foundation Division of Materials Research 1460564), and the Norman Hackerman Advanced Research Program of the Texas Higher Education Coordinating Board (003652-0022-2013)

![](_page_37_Picture_0.jpeg)

![](_page_37_Picture_1.jpeg)

![](_page_37_Picture_2.jpeg)

### ide rylami Properties ບ Content 4 of Independent Thermal

![](_page_37_Figure_4.jpeg)

acrylamide had negligible effect on the melting Copolymerization with temperature

 $\bullet$ 

increasing lauryl acrylate content in The melting point decreased with the copolymer

# Conclusions

- and lauryl temperatures, based on the acrylate exhibited tunable melting stearyl acrylate Copolymers of acrylate content
  - opolymerization with acrylamide had negligible effect on the melting temperature
- exhibited rheological behavior that is typical of without acrylamide polymer melts opolymers

terminal

- dynamic moduli (G', G"), and impacted the terminal region, indicating possible presence of hydrogen bonding acrylamide increased the Addition of
  - Future experiments will explore the impact of UO route to improve the physical properties of hydrogen bonding in these systems mechanical behavior, offering a viable acid-derived polymers fatty

# **Acknowledgements**

thank the I his work is supported by the National Science Foundation (DMR-1611376), including the HCC NSF-REEMS Program (DMP 11275) all participating members of (DMR-1460564). e members of also We group. the all research Robertson researd NSF-REEMS and the REU program. to grateful are

> 5  $\rightarrow \kappa$

Abstract & References QR Code

![](_page_37_Picture_16.jpeg)

![](_page_37_Figure_17.jpeg)

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# Chail Side the teci

- Сh Varying the
  - terminal region behavior

![](_page_37_Picture_30.jpeg)

ш

# **Motivation**

sustainable materials with improved quality while reducing overall costs is crucial in order to overcome the challenges of scarcity of raw and such as petroleum oils. In this project we explore vegetable oils and their fatty derived acids as sources for polymers. We are focusing demand to create innovative on two long chain acrylate monomers de from fatty acids, stearyl and lauryl acrylates. challenges material sources increasing The

Ò 0 Stearyl acrylate Lauryl acrylate

 $\circ = \langle$ 

acrylate exploring the lar interactions, the as derived undergoes Supramolecular such of polymers the properties which the properties of polyn acids, we are exp n of supramolecular copolymerize bonding. We will copolyn vith acrylamide, dynamic hydrogen exhibit dyn hydrogen bonding with the incorporation fatty self-healing. improve monomers as polymers ШO such 2 Ļ

O =Acrylamide:

 $\mathsf{NH}_2$ Project objectives:

- Tune the physical properties of polymers acids through derived from fatty
  - incorporation of supramolecular
    - interactions
- chain length on the polymer properties side Investigate the effect of the

# **Methods** Materials &

Polymers synthesized with reversible addition-fragmentation chain transfer (RAFT) polymerization

polymerization "controlled" technique J RAFT:

Conventional Controlled

- Polymer composition and molecular weight magnetic resonance and gel permeation distribution characterized with nuclear chromatography •
  - Thermal properties explored with differential scanning calorimetry
    - moduli characterized with Dynamic
      - rheology Polymer ( •
- her compositions explored: % lauryl acrylate (relative to total lauryl and stearyl acrylate content) varied from 10-90 mol%
- Acrylamide added at a concentration of 4 wt%

![](_page_37_Picture_54.jpeg)

#### Enhancing Flexibility of Polymer Conductor through Surfactant and Solvent Additives Joshua Jackson<sup>1</sup>, Jorge Wu Mok<sup>2</sup>, and Rafael Verduzco<sup>2</sup>

<sup>1</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College 77082

<sup>2</sup>Department of Biomolecular and Chemical Engineering, Rice University, TX 77005

\*Corresponding author:

Rafael Verduzco, Jorge Wu Mok, Department of Biomolecular and Chemical Engineering, Rice University, 6100 Main St. MS-362, Houston, TX77005 (E-mail: <u>rafaelv@rice.edu</u>)

Organic photovoltaic (OPVs) devices have emerged as a promising source for harvesting renewable energy from sunlight. Despite tremendous advances on the OPV technology, mechanical failure still the limiting factor that prevents the marketability of these devices. Here, we improve the mechanical flexibly and stretch-ability of poly(3,4ethylenedioxythiophene) PEDOT:PSS electrodes by using a combination of ionic additives and solvents. PEDOT: PSS has poor mechanical properties and cracks and fractures occur at minimal tensile strains. Also, when PEDOT: PSS was processed in a thin film we found it has low electrical conductivities. In order to improve both mechanical and electronic properties, we tested the effects of different additives, such as Capstone FS-30 and Zonyl FS-300 for improved flexibility and stretch-ability; and dimethyl sulfoxide (DMSO) for improved conductivities. We prepared blends with various compositions of PEDOT: PSS, DMSO, Zonyl or Capstone. Throughout the research it was shown that with increased concentration of Capstone fs-30, PEDOT: PSS will become more stretchable but less conductive. However, adding DMSO made PEDOT: PSS more conductive but less stretchable. We found that blends with 75 wt.% PEDOT.PSS, 20 wt.% Capstone FS-30 and 5 wt.% DMSO demonstrated significant improvements in mechanical strains and conductivities. In summary, we this work demonstrates that ionic additives can be used to improve the mechanical strains and electronic conductivities of PEDOT: PSS thin films. Future work I want to try to use different additives such as Bis (trifluoroethane) sulfonamide lithium salt and 4-(3-Butyl-1-imidazolio)-1bultanesulfonic acid triflate to test mechanical and electronic improvements.

#### Acknowledgments

This work was supported by the National Science Foundation REU-REEMS program (DMR- 1460564) and the NSF CAREER DMR- 1352099.

![](_page_38_Picture_10.jpeg)

![](_page_39_Picture_0.jpeg)

l exas

![](_page_39_Picture_2.jpeg)

# Conclusion

# on Mylar stretch at 70%**PEDOT.PSS** being

![](_page_39_Picture_5.jpeg)

and d of PEDOT PSS improves the tensile strain up to 40% compared T PSS 11% tensile strain on top of PDMS. We found that blends wt.% PEDOT PSS, 20 wt.% Capstone FS-30 and 5 wt.% DMSO and added to and 5 wt.% DMSO strains ave shown that the more capstone that was PSS improves the tensile strain up to 40% c mechanical added an active layer on top of the placed it on Mylar and we noticed up to a 70% strain. 30 Ц С Capstone improvements .PSS, 20 wt.% significant far have so far ha f PEDOT. conductivities. We DOT **PEDOT.PSS** demonstrated 75 wt.% PE the blend of results 75 with Our **t**0

# Future Work

additives such as Bis (trifluoroethane) sulfonamide 3-Butyl-1-imidazolio)-1bultanesulfonic acid triflate to efficient way to test conductivity in the polymers test mechanical and electronic improvements. different and

Acknowledgements

would like to thank the National Science Foundation for the funding our mentor, Dr. Rafael Verduzco, our supervisor, and Jorge Wu Mok, ou the Chemical and Biomolecular Department at Rice University.

# References

Adam Zhenan Bao." A highly stretchable, transparent, and conductive polymer" Suchol Savagatrup , Esther Chan , Sandro M. Renteria-Garcia , D. Printz , Aliaksandr V. Zaretski , Timothy F. O'Connor , Daniel Liu, Noelle I. Rabiah, Zheng Chen, Jong Won Chung, Linder, Michael F. Toney, Boris Murmann and Zhenan

PEDOT: Principles and Applications of an Intrinsically Conductive Polymer , CRC Press , New York 2011

# <sup>1</sup> Department of Chemical and Biomolecular Engineering, Rice University, Houston, Texas onductor through Surfactant Wu Mok<sup>1</sup>, Rafael Verduzco<sup>1</sup> <sup>2</sup> REEMS program, Houston Community College, Houston, /es ditiv Jorge 12 ZOUN J C L POIV Joshua Jackson<sup>1,2</sup>, Rodrigo ano nhancing Flexibility of

# K EXPERIMEN.

was (PDMS), and Mylar microscope ethyl sulfoxide (DMSO) a com PEDOT:PSS conductivity. This enhancing PEDUI: רשט ערט אושט שישט שישט שישט שישט שוון enhancing PEDUI: Polydimethylsiloxane (PDMS) phthalate (pet)). Stretched under a focused glass,

![](_page_39_Figure_19.jpeg)

- salt Finding an use lithium **t** Try
- We
- Lissel, Jia Christian L  $\sim$  $\overline{}$

# Background 0 **S** Introduction

ause of their flexibility in tuning the molecular s. Their solution process-ability offers additional electronics. Unfortunately, high conductivity and that I have been working with is poly(3,4-ethylenedioxythiophene) (PEDOT: PSS). PEDOT.PSS has poor mechanical properties and cracks and fractures occur at minimal tensile strains harvesting these devices technology source for the OPV t of prevents the marketability OD promising advances σ as tremendous nerged John Marching polymers are good candidates because structures and electrical and mechanical properties. Th advantages for large-scale production of flexible elect high stretchability have not been achieved simultanen that I have home most been achieved simultanen that em factor Despite have limiting devices sunlight. the (OPVs) energy from su l failure is still polymers are photovoltaic mechanical renewable Organic

# S RESU

PEDOT.PSSS with no addictives %

- 95% PEDOT.PSS and 5% **IVNOZ**
- 90% PEDOT.PSS and 10% DMSO
- 85% PEDOT.PSS, 10% Capstone 5% DMSO
- 98% PEDOT.PSS, 1% Capstone, 1% DMSO and
- and apstone 1% C
- 89%PEDOT.PSS, 20%Capstone, and 1% DMSO
- 20%75 % PEDOT.PSS,

![](_page_39_Picture_38.jpeg)

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### study, this placed polar

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![](_page_39_Picture_47.jpeg)

COMMUNITY COLL

HOUSTON

# STRAIN TENSILE

#### **Rodrigo Munoz-Zarruk**

#### **Poster Number Twelve**

**Optimization of thiol-ene coupling for substrates for flexible photovoltaic devices** Rodrigo Munoz-Zarruk<sup>1</sup>, Joshua Jackson<sup>1</sup>, Jorge Wu Mok<sup>2</sup>, and Rafael Verduzco<sup>2,\*</sup>

<sup>1</sup>REEMS REU Program, West Houston Center for Science & Engineering, Houston Community College, Houston, TX 77082
<sup>2</sup>Department of Biomolecular and Chemical Engineering, Rice University, TX 77005

\*Corresponding author:

Rafael Verduzco, Jorge Wu Mok, Department of Biomolecular and Chemical Engineering, Rice University, 6100 Main St, MS-362, Houston, TX 77005 (E-mail: <u>rafaelv@rice.edu</u>)

Organic Photovoltaic (OPV) devices have the ability to harness energy from solar radiation and produce electricity that can be used in a wide variety of applications. As opposed to clunky and relatively hard-to process silicon-based solar cells, OPVs can be easily and relatively inexpensively processed into thin films that exhibit good electrical properties. However, producing OPVs that are flexible is still a challenge when trying to maintain acceptable efficiencies because current polymers used in the (electricity producing) layer of cells do not exhibit high enough (tensile strength modulus) before failing and rendering the cell obsolete. Further, devices are tested on hard substrates with electrodes and interlayers that are brittle and prone to cracking. To address the issue, we propose the use of light-activated chemistries for the fabrication of flexible substrates. Specifically, we focused on the polymerization of a tetra functional thiol, pentaerythritol tetrakis(3-mercaptopropionate) (PETMP), and an allyl vinyl ether, pentaerythritol allyl ether (PAE), using two different catalyzation processes: (UV irradiation) and (thermal annealing). 2-Hydroxy-2-methylpropiophenone (HMPP) is used as a photo-initiator during the UV irradiation polymerization, in which we vary UV exposure times, solution concentration and wavelength type in attempts to optimize processing. After using (ChR Profilometry) to measure thickness in polymer thin films, it was found that five minutes of 300 nanometer UV exposure produces acceptable films with thicknesses ranging from (70-90 nanometer). Azobisisobutyronitrile (AIBN) is used as thermal-initiator during heat-catalyzed polymerization process. The following variables were observed during optimization: thermal annealling time, thermal annealling temperature, exposure to air, polymer solution concentration, and percentage of thermal-initiator used. Films attained acceptable thicknesses, per profilometry results, after thermally annealing 30mg/mL solution of polymer network for 5 minutes at 65C. Results indicate ease of processing of successfully cross-linked

thiol-ene network polymers. These materials provide a simple approach to flexible and solution processible photovoltaic devices.

![](_page_42_Picture_0.jpeg)

# su bstrates for Verduzco <sup>1</sup> Department of Chemical and Biomolecular Engineering, Rice University, Houston, Rafael L O J $\bigcirc$ **—** Vu Mok Orge Ŭ P Munoz-Zarruk<sup>1</sup>,<sup>2</sup> otim zation Rodrigo

![](_page_42_Picture_2.jpeg)

![](_page_42_Picture_3.jpeg)

# **Optimization Results**

coupling through light-activated and thermal annealing es can both be optimized to keep fabrication/manufacturing processes fairly simple and cost effective

UV Irradiation Optimized Process:
UVO cleanse substrates (12 minutes)
Deposit 180 microliters of PETMP:PAE (1:1) solution at 30 mg/mL concentration

Short wave (300 nanomenter) Ultraviolet Irradiation for 1 minutes seconds at 1000 rpm for 1 minute and 30 Spincoat

Thermal Annealing Optimized Process:

UVO cleanse substrates (12 mins) Deposit 180 microliters of PETMP:PAE (1:1) solution at 20 mg/mL concentration

at 1000 rpm for 1 minute and 30 seconds anneal on hot plate at 65C for 5 minutes Spincoat Thermal á

# Work Future

determine **t** polymers thiol-ene network of mechanical properties

and testing devices ΟPV coupled polymers to Incorporating thiol-ene

# Acknowledgements

the would like to thank the National Science Foundation Division of Materials Research (DMR 1460564) and Mr. Bartlett Sheinberg for selecting me to A nart of the Research Experiences and Exploration in Materials and guidance throughout to thank my mentor, Jorge Wu Mok, Exploration and direction Experiences of the Research experience Research (DMR ' be a part of the

Rafael

Dr.

research

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# References

ر من منتقد المالية المالي مالية المالية ال مالية المالية ا ene network-forming systems. Σ and Beaujon, Cook, W. D., Chen, F., Pattison, D. W., Hopson, P. (2007), Thermal polymerization of thiol-ene network Polym. Int., 56: 1572–1579. doi:10.1002/pi.2314 Hoyle, Charles E. and Bowman, Christopher N. (201

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![](_page_42_Figure_19.jpeg)

![](_page_42_Figure_20.jpeg)

REEMS program, Houston  $\sim$ 

# and Introduction

![](_page_42_Figure_24.jpeg)

Instrument: Hot Plate (?

![](_page_42_Figure_31.jpeg)

### Thermal anneal on hot plate with **AIBN** Chloroform (CHCl3) rinse by functional groups Polymer Concentration: AIBN: 3 mg Chlorobenzene: 3mL PETMP: 176.47 mg PAE: 123.35 mg Solution Formula PETMP:PAE (1:1)\* 3 mg 150C 100mg/mL 85C 65C 1% wt at: ... 2) 4)

![](_page_42_Picture_33.jpeg)

HOUSTON COMMUNITY COLLEG

# **Organic Photovoltaic Devices**

- Harvest energy from the Sun and convert it into Low manufacturing costs (R2R production) Thin films allow for countless applications

# Challenge

Flexible solar cells exhibit lower efficiencies

# Solution

Optimizing polymerization of thiol-ene network polymers for use in flexible substrates

http://www

# Irradiation

# 12 mins **Fabrication Process** UVO cleanse 1

- Deposit 180 microliters of  $\widehat{\mathbf{N}}$
- Spincoat 1000 rpm for 1:3( solution  $\widehat{\mathfrak{S}}$ 
  - UV irradiation (Short mins 4
- Chloroform (CHCl3) rinse wave/Long wave (2)

# PETMP:PAE (1:1)\* with **Solution Formula**

\*1:1 by functional groups 5% wt HMPP

# Polymer Concentration: 100mg/mL

Chlorobenzene: 3mL PETMP: 176.47 mg PAE: 123.35 mg 15 mg HHMP:

# **Profilometry** ChR NanoFab Profilometer used to measure Polymer network film thickness

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1,600 -	1,400 -	1 200 -		000	800	000	000	400	000	200	0			
Film Thickness (angstroms)														

			829	▲646				5		
1,400 ⊤	000		1,000	008	009	400		, <del>,</del>		
Film Thickness (angstroms)										

### 2017 REEMS Research Faculty

### Dr. Jakoah Brgoch, Department of Chemistry, University of Houston <u>http://jbrgoch.chem.uh.edu/</u> REEMS REU Students: Gelareh Nobakht and Zeshan Rizvi

Students in the Brgoch group are experts in inorganic synthetic chemistry, numerous characterization techniques, and computational chemistry methods. Connecting these research approaches allows the group to solve current problems in a multitude of complex functional inorganic materials.

![](_page_43_Figure_3.jpeg)

*Front (L-R)*: Shruti Hariyani, Phu-Cuong Phan, Angelica Cobb, Sogol Lotfi, Erin Finley, Adhessha Danthanarayana, Anna Duke, Ya Zhuo *Back (L-R)*: Gayatri Viswanathan, Amber Lim, Aria Mansouri, Anton Oliynyk, Jakoah Brgoch, Sean Bailey

Dr. Margaret S. Cheung-Wyker, Department of Physics, University of Houston and Center for Theoretical Biological Physics, Rice University <u>REEMS REU Student: Frank Kornet</u>

![](_page_44_Picture_1.jpeg)

(Left to right). Front row: Jake, Oleg, Lenaya, Dr. Cheung, Rodney, Pengzhi. Back row: Andrei, Fabio, Victor, Ezzat, Swarnendu.

### Theoretical Biological Physics, Soft Condensed Matter, and Clean Energy https://mynsm.uh.edu/wiki/projects/cheunggroup

One of the goals of the Cheung group is to discover interesting macromolecular dynamics under cell-like conditions by applying molecular simulation methods. Cellular milieu is a crowded and concentrated environment that impacts the behavior of macromolecules. It can affect the rate of protein folding, protein association, and even the overall conformational changes that cannot be probed in dilute solutions. Examples of the simulation tools used in our investigations include coarse-grained molecular simulation, all-atomistic molecular simulation, and bioinformatic data-mining to investigate the structural behavior and statistical properties of large biomolecules in cellular milieu. To tackle macromolecular dynamics across multiple orders of magnitude in both space and time, we develop a state-of-the-art multi-scale molecular simulation and the utilization of high-performance computing resources to simulate very large systems efficiently.

### Dr. Zachary Cordero, Department of Materials Science and NanoEnginering, Rice University <u>https://msne.rice.edu/Content.aspx?id=2147484079</u> REEMS REU Students: Andrew Catalanotto and Nathaniel Ocanas

![](_page_45_Picture_1.jpeg)

Zachary Cordero Assistant Professor

Dr. Cordero leads the Additive Lab in the Department of Materials Science and NanoEngineering at Rice University. The Additive Lab has two complementary research thrusts. The first thrust is to develop metal additive manufacturing techniques for printing parts with complex shapes and precisely-controlled microstructures. The second leverages these techniques to probe structure-property relationships in metals.

Dr. Cordero is an assistant professor of Materials Science and NanoEngineering. He earned both his B.S. in Physics and his Ph.D. in Materials Science and Engineering from the Massachusetts Institute of Technology.In between his undergraduate and graduate programs, Dr. Cordero worked at the Lawrence Berkeley National Laboratory, where he was first exposed to powder metallurgy and metals processing. After receiving his PhD, Dr. Cordero spent one year as a post-doctoral fellow at the Manufacturing Demonstration Facility of the Oak Ridge National Laboratory. There, he developed improved process monitoring, quality control, and microstructure design tools for powder-bed, metal additive manufacturing technologies. Dr. Cordero launched the Additive Lab in the Department of Materials Science and NanoEngineering in July 2016.

![](_page_45_Picture_5.jpeg)

Dr. James K. Meen, Department of Chemistry and The Texas Center for Superconductivity, University of Houston <u>http://www.uh.edu/research/mcf/</u> <u>REEMS REU Students: Brenda Arceneaux, Monica Martinez and</u> <u>Brandon Vu.</u>

### Materials Characterization Facility of the University of Houston

![](_page_46_Picture_2.jpeg)

The Materials Characterization Facility (MCF) is housed in the Houston's Science Center (HSC) building of the University of Houston's Main Campus. We are part of the <u>Texas Center for Superconductivity of the University of Houston (TcSUH)</u>. We specialize in micro-chemical/structural characterization of solid state materials by x-ray and electron beam techniques.

Areas of active research include:

- Conduct Phase Equilibria Studies on Systems Related to High Temperature Superconductors (HTS) Materials.
- Development of High Temperature and High Pressure Experiments for Synthesizing Materials.
- Development of the Analytical Capabilities of Electron Beam Instruments.
- Conduct Studies in Petrology and Geochemistry (mainly igneous systems)

Dr. Megan Robertson, Department of Chemical and Biomolecular Engineering, University of Houston<u>http://robertsongroup.chee.uh.edu/</u> <u>REEMS REU Student: Kevin Scholtes</u>

![](_page_47_Picture_1.jpeg)

#### Polymeric Materials Tailored Structure, Properties, and Function

The objective of our research group is to develop polymeric materials with enhanced physical properties and function. We specialize in polymer synthetic techniques, structural characterization (small-angle neutron, x-ray and light scattering), thermodynamics and self-assembly, and development of structureproperty relationships.

#### Research projects are focused on the following areas:

Sustainable and biodegradable polymers derived from renewable resources Advanced materials for wind energy Structure and dynamics of block copolymer micelles pH-responsive, antifouling polymer brushes Multicomponent and multiphase polymer blends Dr. Laura Smith Callahan, PH.D., Neurosurgery & Center for Stem Cell and Regenerative Medicine

### https://med.uth.edu/neurosurgery/smith-callahan-lab/ REEMS REU Students: Yasaman Adel and Tasmia Nadeem

![](_page_48_Picture_2.jpeg)

The Smith Callahan Laboratory focuses on the developing tissue engineering approaches toward clinical treatments for spinal cord injury, traumatic brain injury and cartilage defects using an interdisciplinary approach involving techniques from cell, molecular, and stem cell biology, chemistry, and material science. Utilizing engineering approaches, the laboratory seeks to optimize scaffold design and the expansion of clinically relevant cell sources.

An Assistant Professor in the Department of Neurosurgery, <u>Dr. Smith Callahan</u> earned her doctorate in Biomedical Engineering from the University of Michigan, where her work under focused on the effects of nanofibrous scaffolding on the osteogenic differentiation of embryonic stem cells. Upon completion of her thesis, she was awarded a post-doctoral fellowship on the Regenerative Science T90 training grant which allowed her to further study the effects of nanofibrous scaffolding on the neural differentiation of embryonic stem cells. To obtain additional training in peptide and polymer chemistry and soft material characterization, Dr. Smith Callahan transitioned to a post-doctoral position at the Institute of Polymer Science at the University of Akron with Matthew L. Becker. At the University of Akron, her work focused on the effects of bioactive peptides and gradient hydrogels on stem cell differentiation to mesenchymal and neuronal lineages

#### **CURRENT PROJECTS**

 Development of multi-component scaffolds to facilitate tissue regeneration through better replication of the native extracellular matrix.
 Optimization of culture surfaces for the differentiation of human induced pluripotent stem cells to neural stem cells and oligodendrocyte progenitor cells.
 Identification of optimal artificial matrix properties such as bioactive signaling moiety concentration or mechanical properties using combinatorial approaches.
 Synthesis of novel biomaterials for spinal cord, brain, and vertebral disk repair.

![](_page_48_Picture_7.jpeg)

### Dr. Rafael Verduzco, Department of Chemical and Biomolecular Engineering, Rice University http://verduzcolab.blogs.rice.edu/

<u>REEMS REU Students: Rodrigo Munoz-Zarruk and Joshua Jackson.</u> The Verduzco laboratory focuses on the development of complex polymeric materials. We take advantage of advanced polymer synthesis techniques and nanoscale characterization tools to design and characterize polymers at multiple length scales. Current areas of interest include all-conjugated block copolymers for photovoltaics, bottlebrush polymers as responsive surface coatings, liquid crystal elastomers for biomedical applications, and polymers for enhanced oil recovery. The unifying theme of this work is engineering materials at the molecular level to achieve a stronger fundamental understanding of material properties.

![](_page_49_Picture_2.jpeg)

### 2017 REEMS REU JUDGING PANEL

#### Dr. Forrest J. "Jack" Agee

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Dr. Forrest J. "Jack" Agee, a member of the Senior Executive Service, is Director of Physics and Electronics, Air Force Office of Scientific Research, Arlington, Va. He is responsible for the \$80 million Air Force basic research program in physics and electronics, assuring the excellence and relevance of a broad research portfolio. His program encompasses hundreds of university grants and supports work of undergraduates and more than 300 graduate students. This work also supports basic research within the Air Force Research Laboratory, industry and overseas. The research includes novel space-craft engineering, semiconductor device research, nanotechnology, electronic sensors,

polarimetry research, gravitometry, lasers, plasmas, high-power radio frequency sources, high-temperature superconductivity engineering, atomic and nuclear physics, space optics, and imaging and opto-electronics. Agee plans, coordinates and executes a research program conducted by scientists in academia, industry and Air Force laboratories. He was appointed to the SES in 1998.

Prior to beginning his career with the Air Force, Dr. Agee worked as a U.S. Navy physicist in acoustics related to silencing submarines. He also worked in low-temperature physics for the U.S. Army at Fort Belvoir, Va. In the 1970s, he joined the Harry Diamond Laboratory in Adelphi, Md. He handled many assignments of increasing scope and responsibility, initially in nuclear electromagnetic pulse hardening and testing for the Safeguard Anti-Ballistic Missile System and for Army tactical systems. As the Technical Director, he led a major, strategic, Defense Nuclear Agency Electromagnetic Pulse Test for the Commander in Chief of Pacific Forces. For three years he managed strategic command, control and communications programs at The BDM Corp., including the Airborne Command Post. In 1982 Dr. Agee became Director of the Aurora Radiation Test Facility that tested the Peacekeeper missile and other systems at the Harry Diamond Laboratory. In the Army's high-power microwaves program, Dr. Agee led research and development efforts in sources, and in 1990, became Director of the Army program. His efforts led to developing the AN/VLQ-9 and AN/VLQ-10 Shortstop electronic warfare systems during operations Desert Shield and Desert Storm. The Shortstop System protected U.S. forces in Bosnia and now has seven versions, including two that are deployed in Iraq in a counter-IED role. Dr. Agee formed the Joint Directors of Laboratories Panel on Directed-Energy Weapons in 1990, and chaired the panel until 1993. That same year, he began his Air Force career in the Directed-Energy Program at Phillips Laboratory at Kirtland Air Force Base, N.M., where he led the research program in high-power microwave sources.

Dr. Agee's principal fields of interest are in electrical engineering, pulsed power, microwave generation, electromagnetics and superconductivity. In physics, his interests include nuclear physics, plasma physics, semiconductor physics and lasers. He has written or co-written more than 200 publications and inventions in these areas.

#### **EDUCATION**

1963 Bachelor of Science in physics, high honors, Clemson University, Clemson, S.C. 1965 Master of Science degree in physics, University of Virginia, Charlottesville 1967 Doctor of Philosophy degree in physics, University of Virginia, Charlottesville

#### **Dr. Andrew Bean**

![](_page_52_Picture_1.jpeg)

Dr. Bean is Professor in the Department of Neurobiology and Anatomy at UTHealth McGovern Medical School and Associate Dean in the MD Anderson-UT Health Graduate School of Biomedical Sciences (GSBS). His laboratory has been studying the molecular mechanisms of membrane protein sorting and endosomal trafficking for 20 years. The Bean Laboratory has used molecular, biochemical, genetic, and cell biological approaches to make novel discoveries concerning the mechanisms of membrane trafficking events that have contributed to understanding the role of mutations that alter trafficking events in cancer and neurological disease. In the GSBS, Dr. Bean manages student recruiting, orientation, admissions, and career development.

### Mr. Zane Marek

![](_page_52_Picture_4.jpeg)

Zane Marek graduated from Texas State Technical College in 1992 with a degree in Electronic Engineering Technology. He began working with JEOL USA as a Field Service Engineer in 1992, assigned to semiconductor equipment (SE) solutions. Zane went back to school while working in the Field Application's Group of JEOL's SE division and received a Bachelor of Science in Business/Marketing from the University of Phoenix. He currently has dual responsibilities as JEOL USA's SE Product Manager and Senior Sales Manager for JEOL's Electron Microscopes.

![](_page_52_Picture_6.jpeg)

### JAMES K. NELSON, JR., PH.D., P.E., C.ENG., F.ASCE

![](_page_53_Picture_1.jpeg)

Dr. James K. Nelson received a Bachelor of Civil Engineering degree from the University of Dayton in 1974. He received the Master of Science and Doctor of Philosophy degrees in civil engineering from the University of Houston. During his graduate study, Dr. Nelson specialized in structural engineering. He is a registered professional engineer in three states, a Chartered Engineer in the United Kingdom, and a fellow of the American Society of Civil Engineers. He is also a member of the American Society for Engineering Education and the SAFE Association. Prior to receiving his Ph.D. in 1983, Dr. Nelson worked as a design engineer in industry and taught as a lecturer at the University of Houston and Texas A&M University at Galveston. In industry he was primarily involved in design of floating and fixed structures for

the offshore petroleum industry. After receiving his Ph.D., Dr. Nelson joined the civil engineering faculty at Texas A&M University. He joined the civil engineering faculty at Clemson University in 1989 as Program Director and founder of the Clemson University Graduate Engineering Programs at The Citadel and became Chair of Civil Engineering in 1998. While at Clemson he received the Award for Faculty Excellence.

In July 2002, Dr. Nelson joined the faculty at Western Michigan University as Chair of Civil and Construction Engineering. At Western Michigan he started the civil engineering undergraduate and graduate degree programs and also chaired the Departments of Materials Science and Engineering and Industrial Design. In May 2005 he joined the faculty at The University of Texas at Tyler. At UT Tyler he was the founding chair of the Department of Civil Engineering and instituted the bachelor's and master's degree programs. In 2006 he became the Dean of Engineering. While serving as Dean, enrollments in the college grew over 250 percent. He returned to Texas A&M in January 2016 and currently serves as the Director of Special Academic Initiatives for the Texas A&M University System.

Dr. Nelson's primary technical research interest is the behavior of structural systems. For over 25 years he has been actively involved in evaluating the behavior of free-fall lifeboats and the development of analytical tools to predict that behavior. His research has formed the basis for many of the regulations of the International Maritime Organization for free-fall lifeboat performance. Since 1988, Dr. Nelson has served as a technical advisor to the United States Delegation to the International Maritime Organization, which is a United Nations Treaty Organization. In that capacity, he is a primary author of the international recommendation for testing free-fall lifeboats and many of the international regulations regarding the launch of free-fall lifeboats. In 1996 Dr. Nelson received the United States Marine Safety Award for accomplishments furthering the cause of safety in the marine field.

He has authored many technical papers that have been presented in national and international forums and co-authored three textbooks. Dr. Nelson chaired a national committee of the American Society of Civil Engineers for curriculum redesign supporting the civil engineering body of knowledge. He is actively engaged in developing strategies for enhancing the STEM education pipeline in Texas and nationally, and has testified before the Texas Senate and House Higher Education Committees in that regard. He chaired the councils for the Texas Higher Education Coordinating Board developing statewide articulation compacts for several engineering and science programs. He chaired the Engineering Field of Study Committee for the Coordinating Board. Dr. Nelson was the primary architect of the UT Tyler Houston Engineering Center through which UT Tyler's engineering programs are available to students in Houston. He also served on the Texas State Board of Education committee preparing the standards for career and technical education. In addition, Dr. Nelson served as the chair of the academic advisory committee for the Texas Board of Registration for Professional Engineers, and chaired the task committee to enhance faculty licensure. He also served as the representative of ASEE on the NCEES Engineering Education Task Force.

### **George M. Stancel, Ph. D.** Senior Vice President, Academic and Research Affairs, UTHealth

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Dr. George M. Stancel currently serves as the Senior Vice President for Academic and Research Affairs at The University of Texas Health Science Center (UTHealth), a position he has held since 2011.

Stancel came to The University of Texas Health Science Center at Houston in 1972 as an assistant professor of Pharmacology at the Medical School. He has assumed many leadership roles before becoming Executive Vice President for Academic and Research Affairs, including president of the Medical School Faculty Senate, president of the Graduate School Faculty, Chairman of the Pharmacology Department at the Medical School, Associate Dean for Education and Research at the Medical School, Executive Vice President for Research of the Health Science Center, Dean of the Graduate School

of Biomedical Sciences, and has served on innumerable committees of the Medical School, Graduate School, and Health Science Center. Stancel notes that he has taught every student who has gone through the Medical School here in Houston and has taught at all six UTHealth Schools and MD Anderson during his tenure. He is especially proud of teaching awards he has received. He has also been active in national professional organizations dedicated to biomedical science and medical education. In addition to being the SVP of Academic and Research Affairs, he is currently Professor of Integrative Biology and Pharmacology at the UT Medical School and Professor of Gynecologic Oncology at MD Anderson (adjunct) and continues to teach pharmacology each year to the medical class.

Stancel's training and research programs have received over \$15 million in grants from the National Institutes of Health (NIH) and other sources. His past research has been on the effects of estrogens and related hormones and drugs on the female reproductive system and their role in hormone related cancers although he no longer has his own active research lab since he focuses now on administrative leadership. He has served on a number of research advisory panels and review groups for The National Institutes of Health (NIH) and other professional organizations. He remains actively involved in teaching and research training of graduate students and postdoctoral fellows in UTHealth training programs and those of the Gulf Coast Consortium. He has published over 200 research articles, scientific abstracts, book chapters, and other works.

Stancel was raised in Chicago, and earned a B.S in chemistry from the College of St. Thomas in St. Paul, MN in 1966. He went on to receive a doctorate in biochemistry from Michigan State University in 1970 and did postdoctoral work in physiology at the University of Illinois at Champaign-Urbana for two years. He is married to Mary Lee (Wiepking) Stancel, and they have three children. Mary taught Spanish and English as a Second Language at the high school level for almost 10 years and then taught pre-school at Bellaire United Methodist's School for Little Children for over 30 years where she was selected as the Outstanding Pre-School Teacher of the Year in Houston by the Martel Foundation in 2004. Stancel has been active in scouting and youth sports and his current hobbies include fishing and biking.

#### Dr. Ming Tang

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Dr. Tang is an assistant professor of Materials Science and NanoEngineering at Rice University. He obtained his Ph.D. in Materials Science and Engineering at MIT. Before joining Rice, he worked at the Lawrence Livermore National Laboratory and also Shell Technology Center, Houston. His group is interested in materials phenomena at mesoscale, which bridge between atomistic building blocks and macroscopic properties. Current research topics include microstructure evolution phenomena in energy storage materials, rational design of microstructure in battery electrodes, morphological evolution and control during 2D materials growth and selfassembly process in emulsion systems. Dr. Tang is planning to join the REEMS REU program in the spring of 2018.

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#### Mr. John Vasselli

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John J. Vasselli is the Dean of the Houston Community College Engineering Center of Excellence. He possesses over forty years of engineering experience ranging from the Federal research environment, through large corporations, to starting and owning his own engineering businesses. John's career as a scientist, researcher, product developer, and serial entrepreneur gives him a full-spectrum understanding of how to turn a new idea into a global corporation. Three of the companies that John led grew to over \$100M/year in annual sales. He holds several patents, and degrees in electrical, biomedical and systems engineering.

#### PROFESSIONAL SUMMARY

Houston Community College System – Dean, Engineering Center of Excellence

- University of Texas Director, Houston Engineering Center and Executive Director of TxAIRE Research Institute
- Carrier Corp. Chief of Technology for Indoor Air Quality & Global Technology Fellow
- Syracuse Center of Excellence for Environmental and Energy Systems Exec. VP
- Green Star Technologies Founder & President
- Houston Advanced Research Center President & CEO
- Syracuse Research Corporation, Director of Corporate Development
- Science Applications International Corporation Corp. Vice President
- Farmers Mills Technologies Founder & President
- Air Force Research Laboratory & DARPA Program Manager Sr. Research Engineer
- United States Air Force Captain

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#### Dr. Marenda A. Wilson- Pham, Ph. D.

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Assistant Dean, Diversity and Alumni Affairs The University of Texas MD Anderson Cancer Center UTHealth Graduate School of Biomedical Sciences

Dr. Wilson-Pham received her BS in Biology at Dillard University in 2002 and her PhD in Microbiology and Molecular Genetics at The MD Anderson UTHealth Graduate School of Biomedical Sciences (GSBS) in 2007 in the laboratory of Dr. Ambro vanHoof. Her research addressed the understanding the molecular mechanisms of nonstop mRNA metabolism. Following her doctoral research, she accepted her first postdoctoral fellowship in the Biochemistry and Molecular Biology Department at The

UT MD Anderson Cancer Center in the laboratory of Dr. Sharon Y.R. Dent. Her research was focused on understanding how histone modifying complexes regulate gene expression. Her second postdoctoral fellowship, in the Molecular and Human Genetics Department at Baylor College of Medicine in the laboratory of Dr. Grzegorz Ira, was focused on the role of DNA double-strand breaks and repair in the maintenance of genome stability.

She joined the GSBS Deans' Office in 2013 to assist in improving the recruitment and retention rates of underrepresented minority students. She is involved in yearly recruitment activities through attendance of conferences and institutions that serve large populations of minority students. Her retention efforts are focused on providing educational and community support to currently enrolled minority students at the GSBS through monthly lunch-and-learn seminars, workshops and mentorship support. She oversees support to all student organizations and alumni networking initiatives of the graduate school.

### **REEMS & West Houston Center for Science &** Engineering Staff

#### Mr. Bartlett (Bart) Sheinberg

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REEMS Principal Investigator & Center Director, West Houston Center for Science & Engineering

### Mrs. Mary Beth Hurd

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**REEMS Administrator** 

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### **REEMS & West Houston Center for Science & Engineering Staff**

#### Dr. Yibran Perera Mercado

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**REEMS Program Manager** 

#### Dr. Gizelle Davis

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HCC Biology Faculty REEMS Student Mentor

Dr. Amanda Hackler

![](_page_61_Picture_8.jpeg)

**REEMS External Evaluator** 

### The West Houston Center for Science & Engineering Advisory Council Members

Dr. Forrest J. (Jack) Agee, Ph.D.

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### WHC Council Chair Chief Executive Officer, ADAMCO Inc.

Mr. Ron Kelley

![](_page_62_Picture_5.jpeg)

WHC Deputy Council Chair President, Strategic Partners, Inc. Dr. Carolyn A. Nichol, Ph.D.

![](_page_63_Picture_1.jpeg)

Faculty Fellow in Chemistry, Director of the Office of STEM Engagement Rice University

Dr. Alan J. Jacobson, Ph.D.

![](_page_63_Picture_4.jpeg)

Director, Texas Center for Superconductivity University of Houston

![](_page_63_Picture_6.jpeg)

### Mr. Robert K. (Bob) Ehrmann

![](_page_64_Picture_1.jpeg)

Managing Director Pennsylvania State Center for Nanotechnology Education & Utilization

![](_page_64_Picture_3.jpeg)